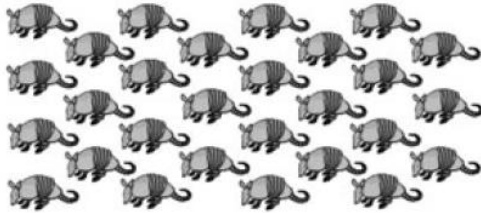


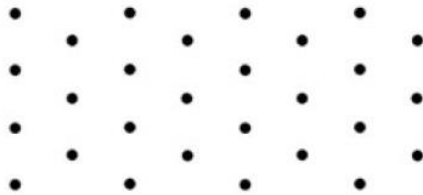
Introdução

Periodic Structure



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Lattice

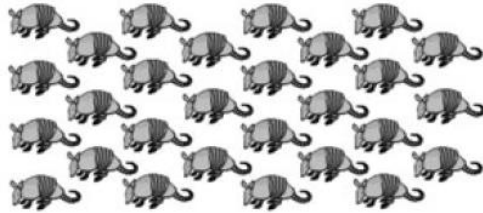


× Repeating Object



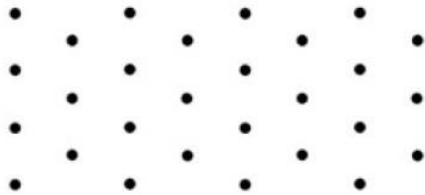
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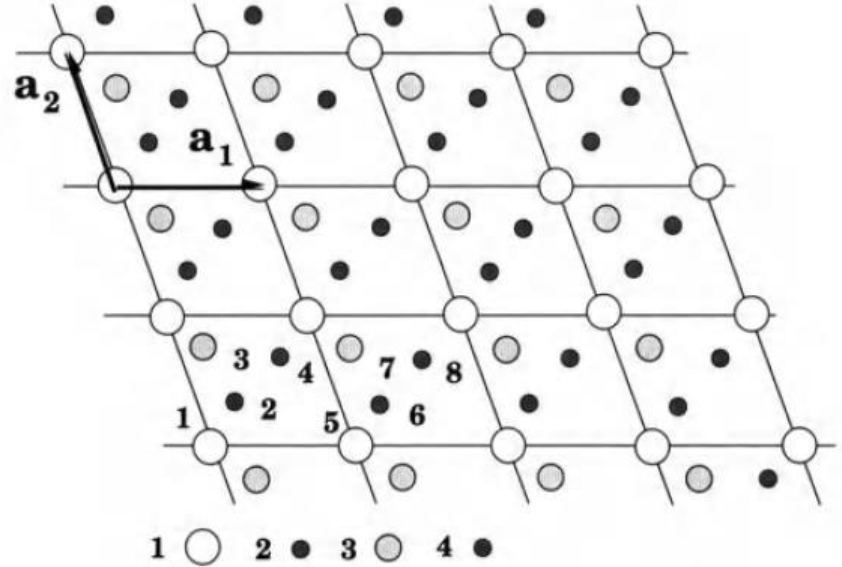


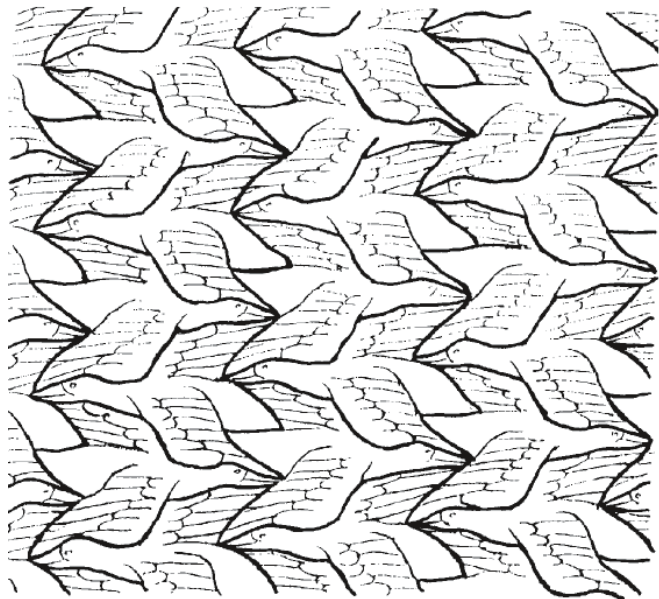
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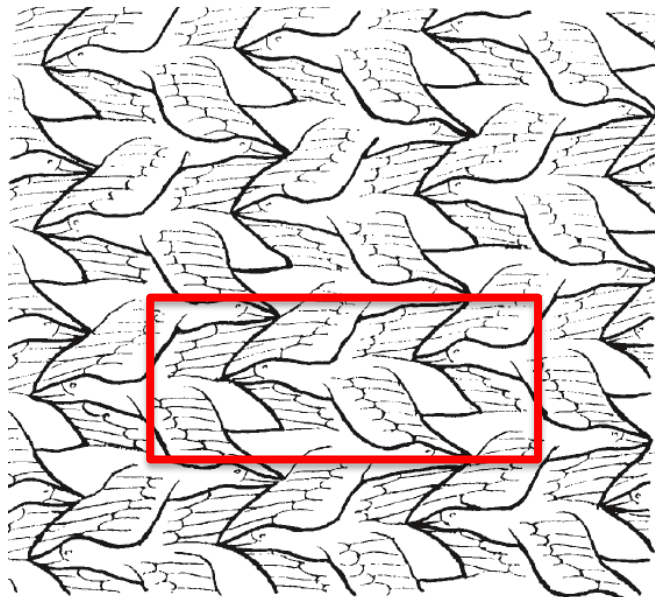
Lattice

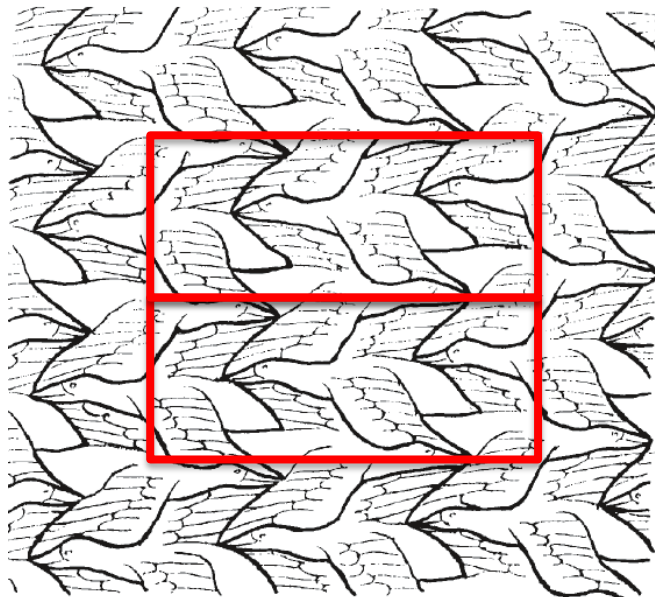


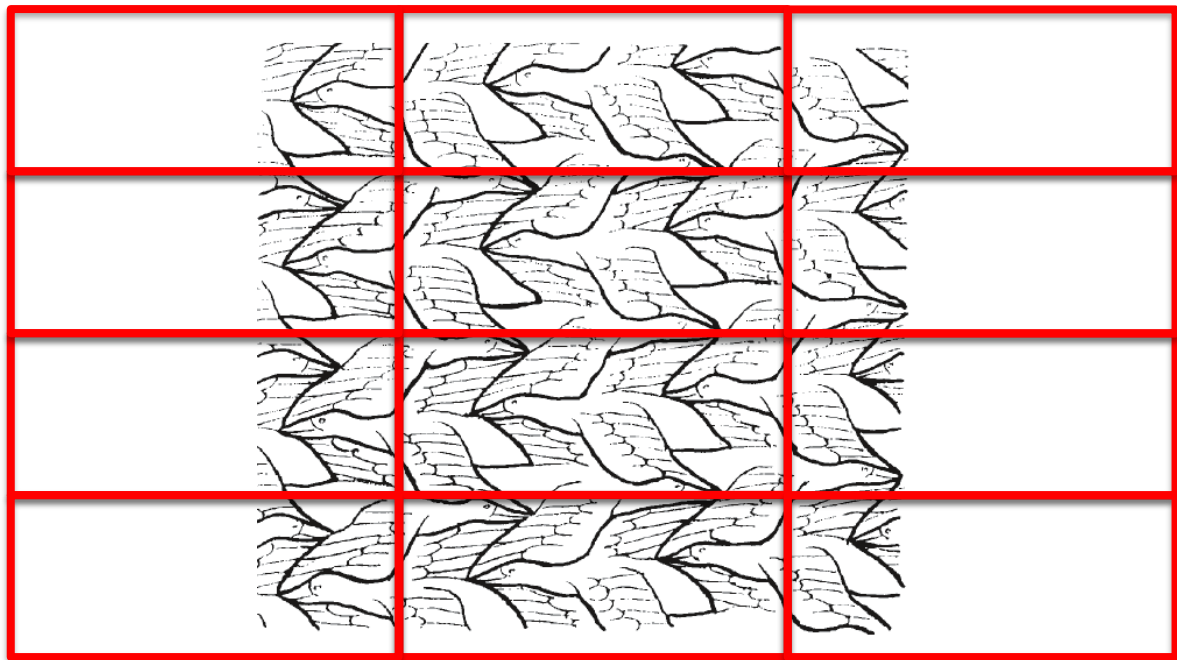
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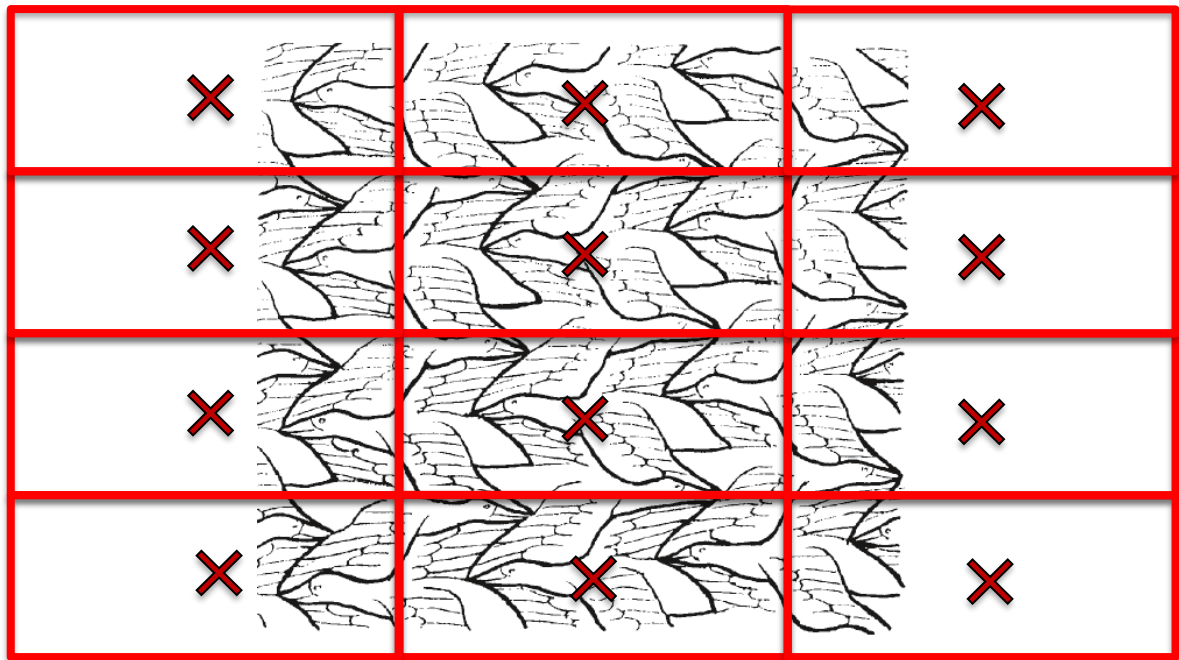


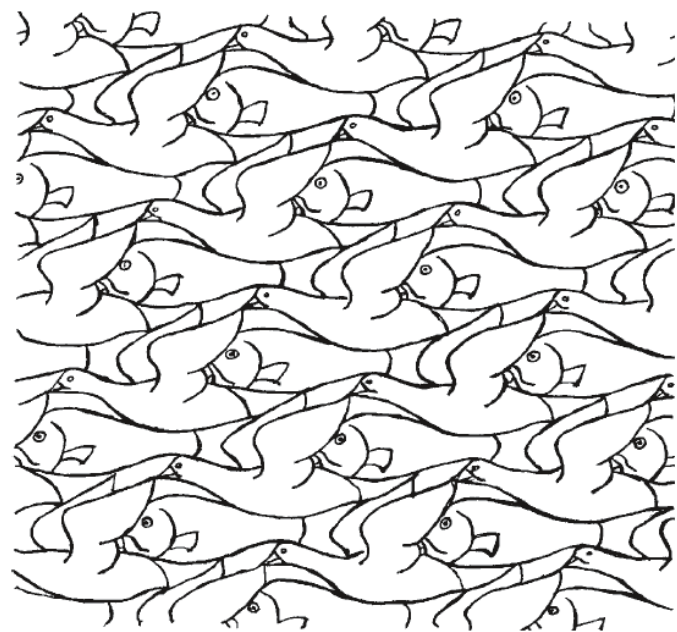


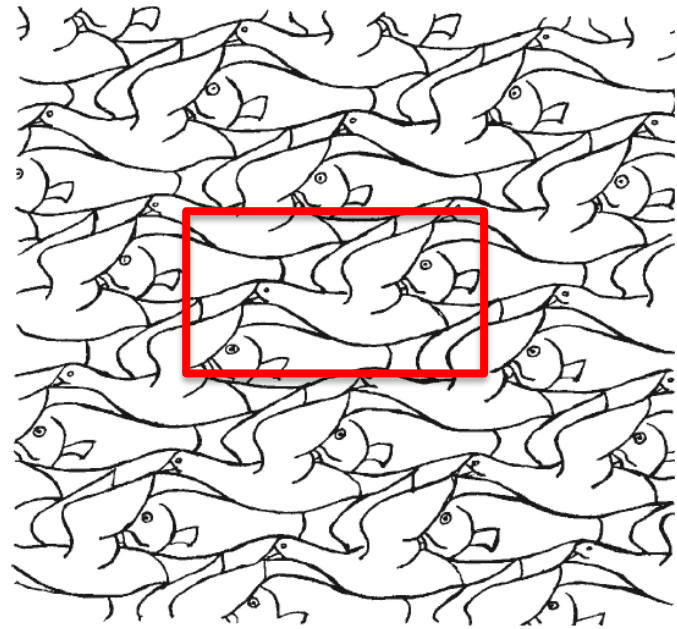


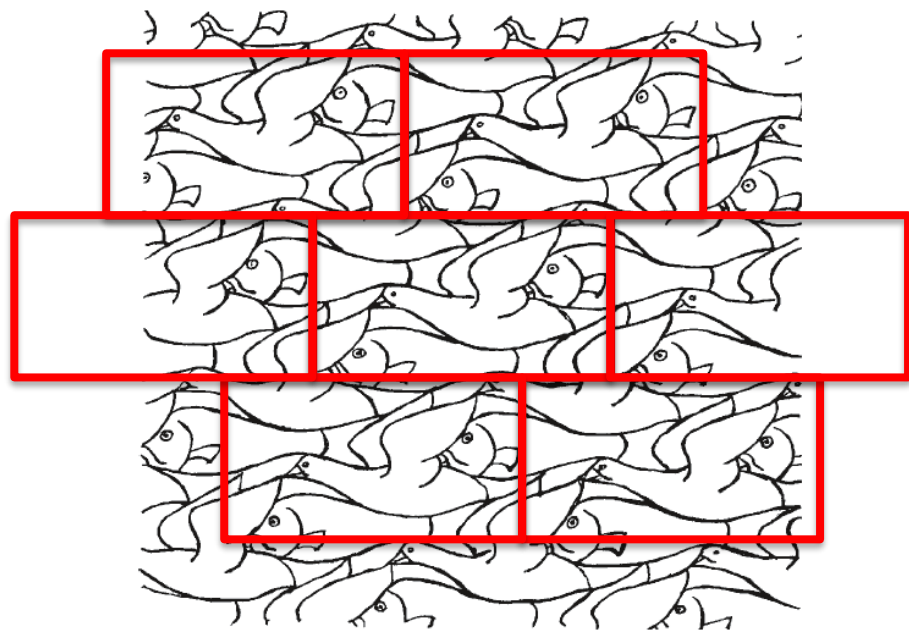
















Definição de Rede de Bravais

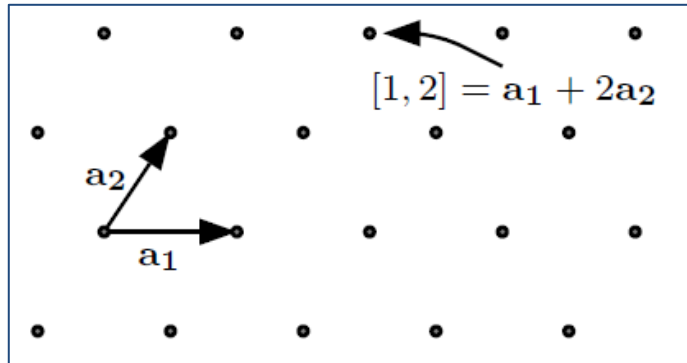
Definição: Uma *rede de Bravais* é um conjunto infinito de pontos definidos pela soma de um número inteiro de *vetores de rede primitivos*.

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$$\mathbf{R}_{[n_1 \ n_2]} = n_1 \mathbf{a}_1 + n_2 \mathbf{a}_2 \quad n_1, n_2 \in \mathbb{Z}$$

$$\mathbf{R}_{[n_1 \ n_2 \ n_3]} = n_1 \mathbf{a}_1 + n_2 \mathbf{a}_2 + n_3 \mathbf{a}_3 \quad n_1, n_2, n_3 \in \mathbb{Z}$$

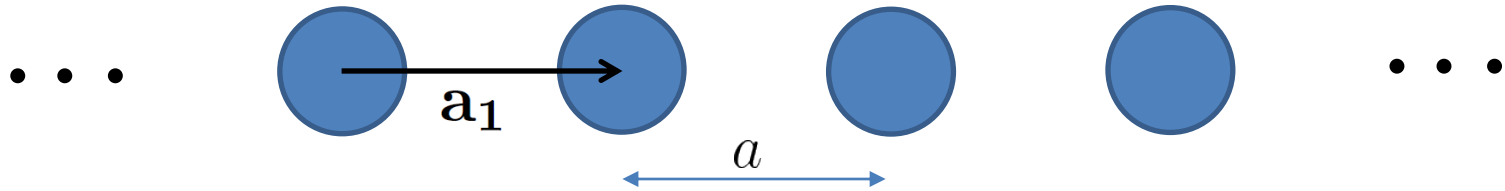


Definição de Rede de Bravais

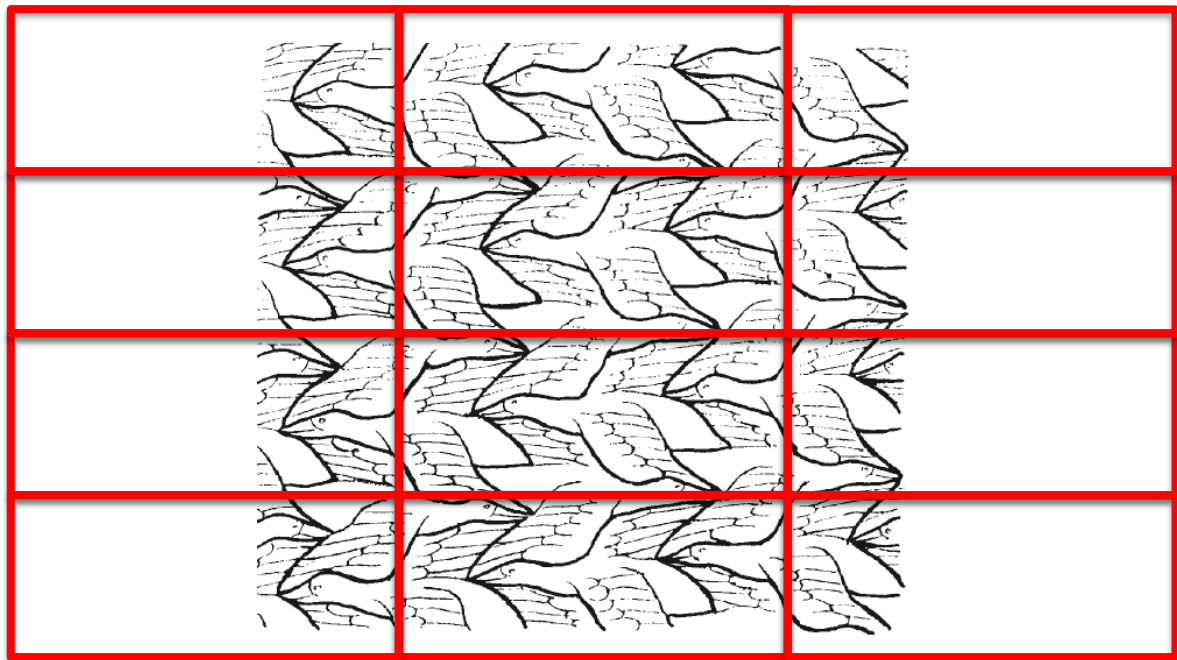
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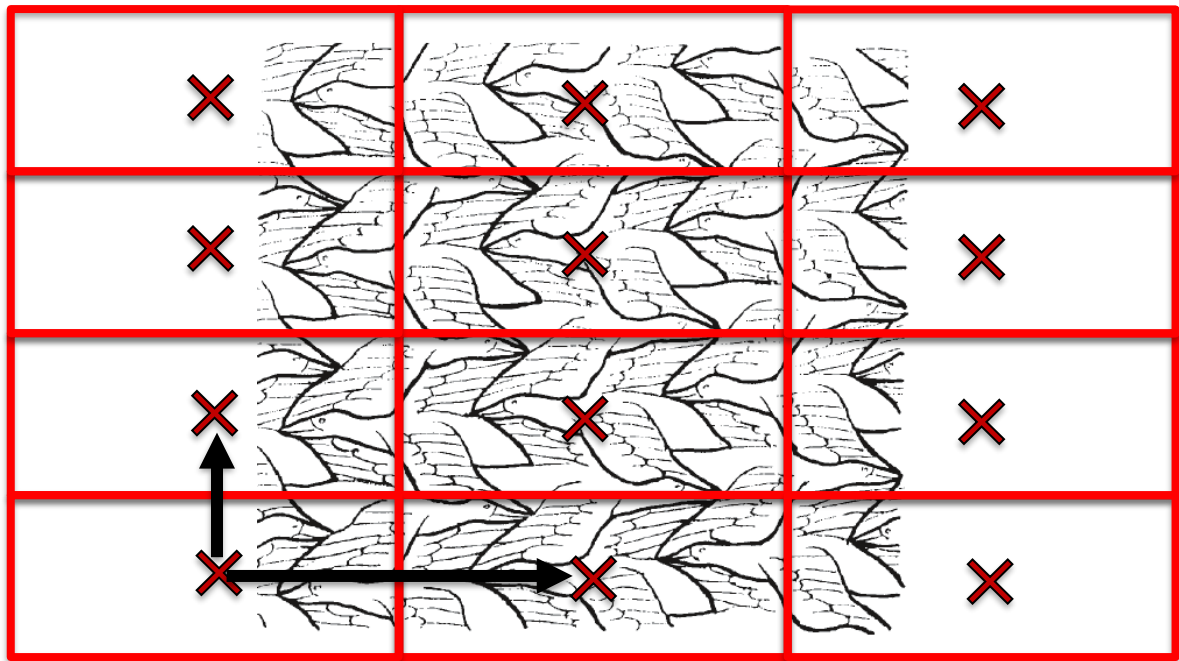
$$\mathbf{R}_{[n_1 n_2]} = n_1 \mathbf{a}_1 + n_2 \mathbf{a}_2 \quad n_1, n_2 \in \mathbb{Z}$$

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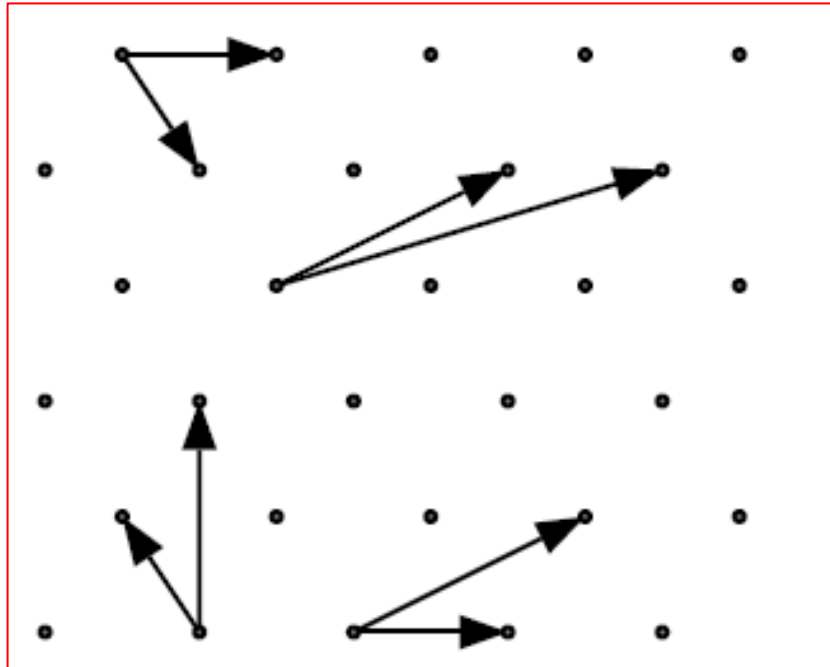
Notem que temos simetria de translação: Seja $f(\mathbf{R})$ um observável da meu cristal, então $T(\mathbf{R}') f(\mathbf{R}) = f(\mathbf{R} + \mathbf{R}') = f(\mathbf{R})$, com $T(\mathbf{R})$ sendo o operador translação





Rede de Bravais

- A escolha dos vetores primitivos (que geram a rede) não é única.



Importante:

CRISTAL = Rede de Bravais + Base

$\mathbf{R}_{[n_1 n_2 n_3]} = n_1 \mathbf{a}_1 + n_2 \mathbf{a}_2 + n_3 \mathbf{a}_3$ Posição das células unitárias na rede de Bravais

As estruturas que se repetem



Importante:

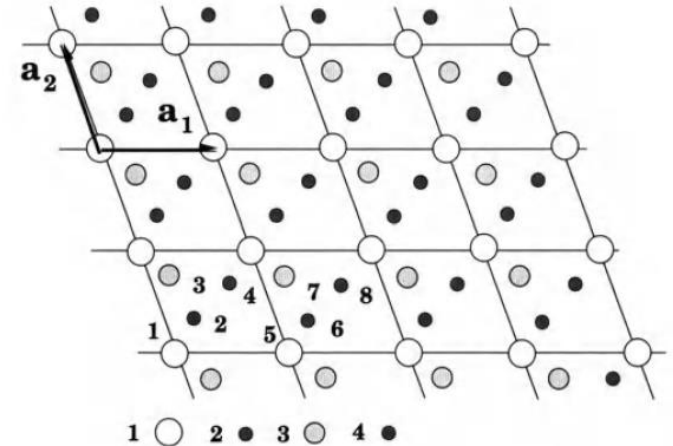
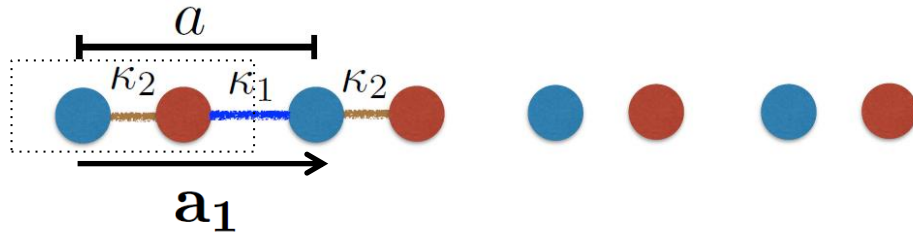
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As estruturas que se repetem

$\mathbf{R}_s = s_1 \mathbf{a}_1 + s_2 \mathbf{a}_2 + s_3 \mathbf{a}_3$ Posição relativa dos sítios da base

$\mathbf{R}' = n_1 \mathbf{a}_1 + n_2 \mathbf{a}_2 + n_3 \mathbf{a}_3 + \mathbf{R}_s$ Posição dos sítios na rede de Bravais



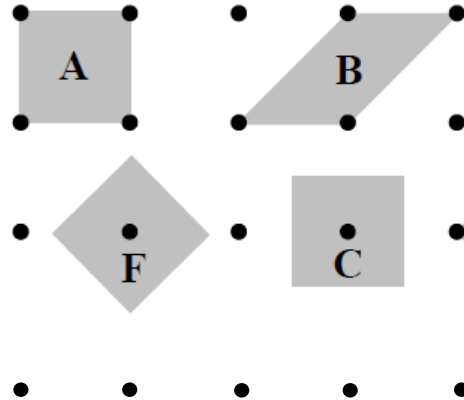
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Definição de Célula Unitária: É uma região do espaço que quando deslocada por um subconjunto dos vetores da rede de Bravais reproduz a rede completa sem sobreposições.

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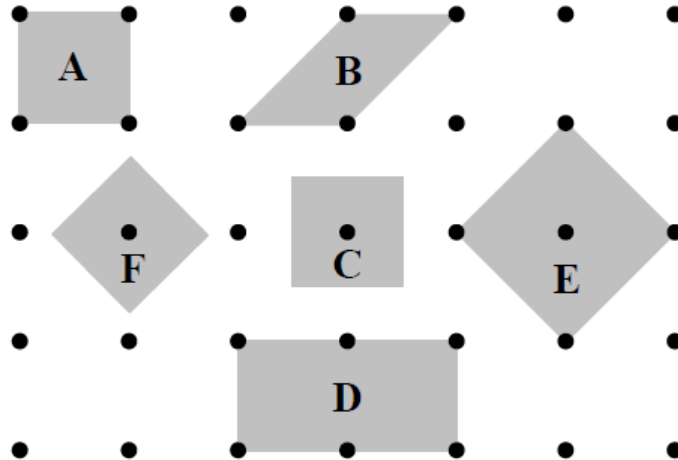
- **Célula Unitária Primitiva:** É uma célula unitária de volume mínimo, i.e. que contém apenas um ponto da rede de Bravais.



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- **Célula Unitária Primitiva:** É uma célula unitária de volume mínimo, i.e. que contém apenas um ponto da rede de Bravais.
- **Célula Unitária Convencional:** É uma célula unitária cujo volume é um múltiplo da célula primitiva, i.e. que contém mais de um ponto da rede de Bravais

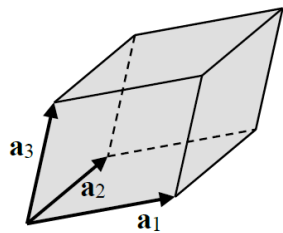


Há infinitas células primitivas, podemos obter cada uma a partir de outra ao cortarmos ela e transladarmos os pedaços por diferentes vetores da rede de Bravais.

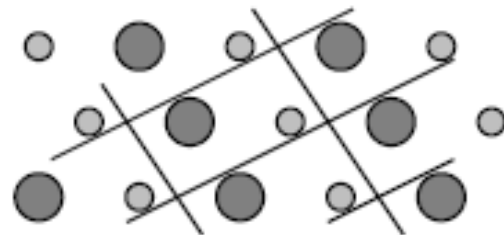
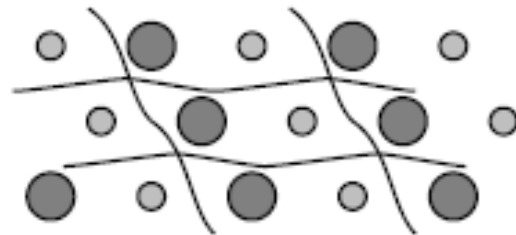
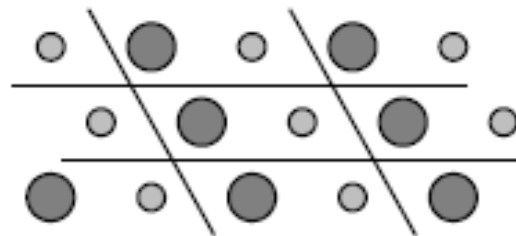


A forma da célula unitária não fornece uma classificação das redes de Bravais!

Volume de uma célula primitiva:

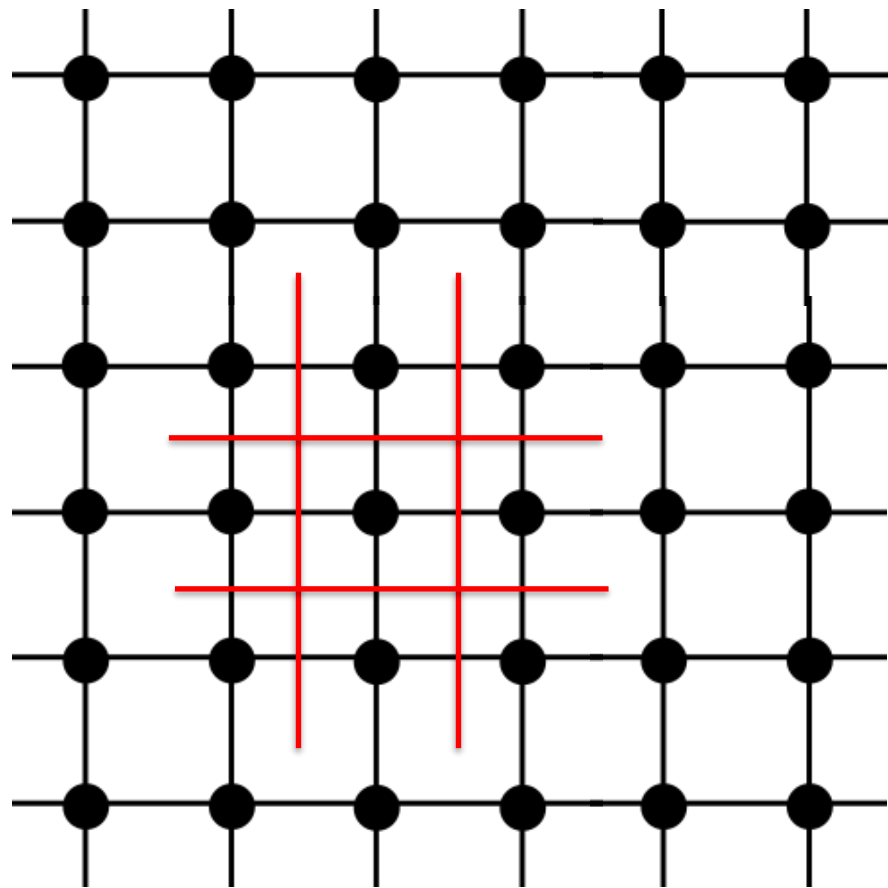


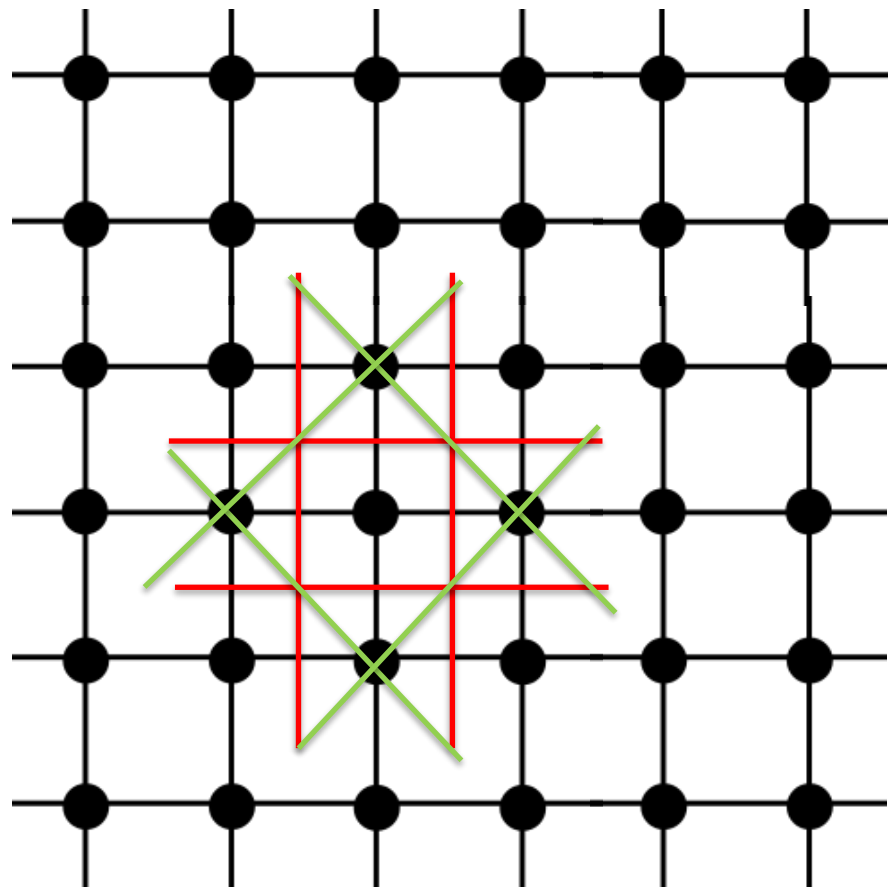
$$v = \left| \mathbf{a}_3 \cdot (\mathbf{a}_1 \times \mathbf{a}_2) \right|$$

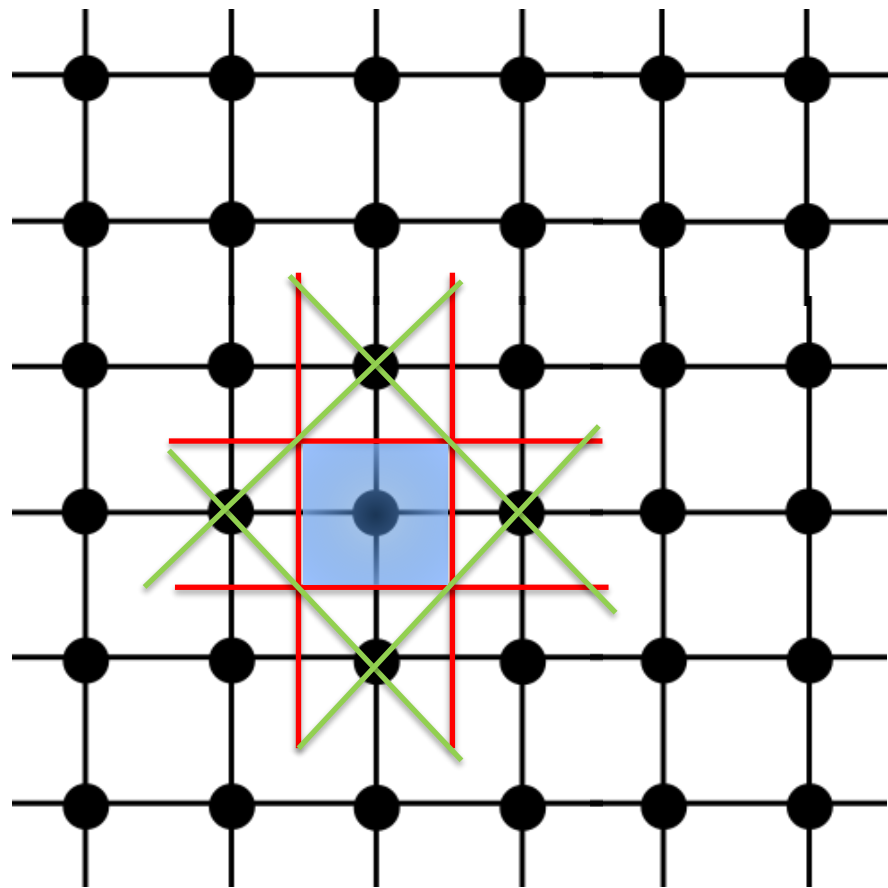


Rede de Bravais

Definição de Célula de Wigner-Seitz: É uma célula primitiva construída no entorno de um dado ponto da rede, cuja região do espaço por ela delimitada é mais próxima deste dado ponto do que de qualquer outro ponto da rede. Na prática, é possível construir a célula de Wigner-Seitz a partir de planos (em redes 3D) ou linhas (em redes 2D) bisecantes às linhas que ligam o ponto em análise e seus vizinhos.

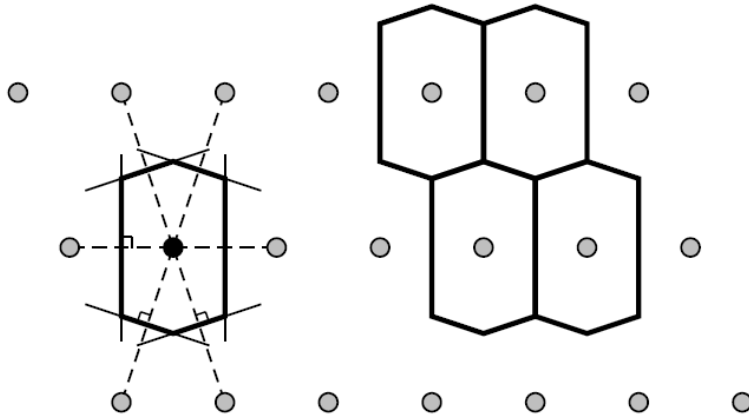






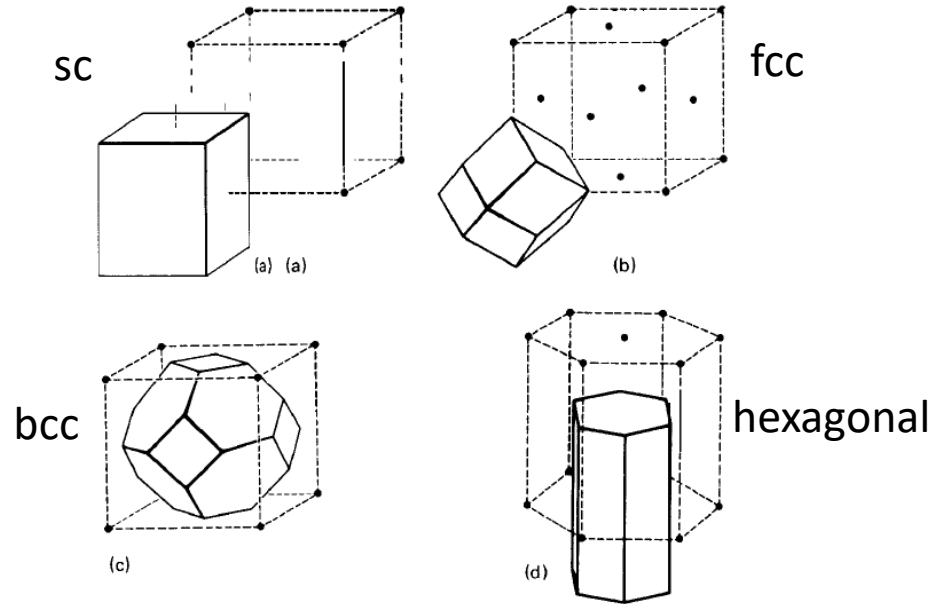
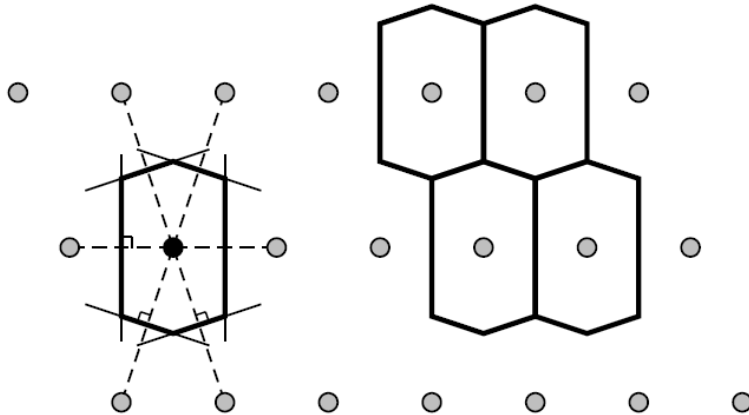
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Classificação das redes de Bravais

Quantas redes de Bravais distintas podemos formar em duas dimensões ou em 3 dimensões?

Para responder essa pergunta, é preciso compreender que as operações de simetria de uma rede pertencem ao que chamamos de *grupo de simetria espacial*.

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O que é um grupo? É um conjunto dotado de uma operação de multiplicação, cuja operação em dois elementos quaisquer leva em outro elemento também pertencente ao conjunto. Este conjunto deve conter um elemento neutro e um inverso.

$$A_i \times A_j = A_k \quad (\text{multiplicação – associativa})$$

$$E \times A_i = A_i \times E = A_i \quad (\text{Elemento neutro})$$

$$A_i \times A_i^{-1} = A_i^{-1} \times A_i = E \quad (\text{inverso})$$

Em geral, $A_i \times A_j \neq A_j \times A_i$

Classificação das redes de Bravais

Exemplo: Grupo de translação



$$\hat{g}(\mathbf{L})\mathbf{R} := \mathbf{R} + \mathbf{L}$$

$$\left\{ \begin{array}{l} \hat{g}(\mathbf{L}_1)\mathbf{R} = \mathbf{R} + \mathbf{L}_1 \\ \hat{g}(\mathbf{L}_2)\mathbf{R} = \mathbf{R} + \mathbf{L}_2 \end{array} \right.$$

$$\hat{g}(\mathbf{L}_2)\hat{g}(\mathbf{L}_1)\mathbf{R} = \hat{g}(\mathbf{L}_2)(\mathbf{R} + \mathbf{L}_1) = (\mathbf{R} + \mathbf{L}_1) + \mathbf{L}_2 \equiv \hat{g}(\mathbf{L}_{12})\mathbf{R}$$

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$$\left\{ \begin{array}{l} \hat{g}(\mathbf{L}_2)\hat{g}(\mathbf{L}_1) = \hat{g}(\mathbf{L}_{12}) \quad (\text{multiplicação}) \\ \hat{g}(E) = \mathbf{I} \quad (\text{Elemento neutro}) \\ \hat{g}^{-1}(\mathbf{L}) := \hat{g}(-\mathbf{L}) \quad (\text{inverso}) \end{array} \right.$$

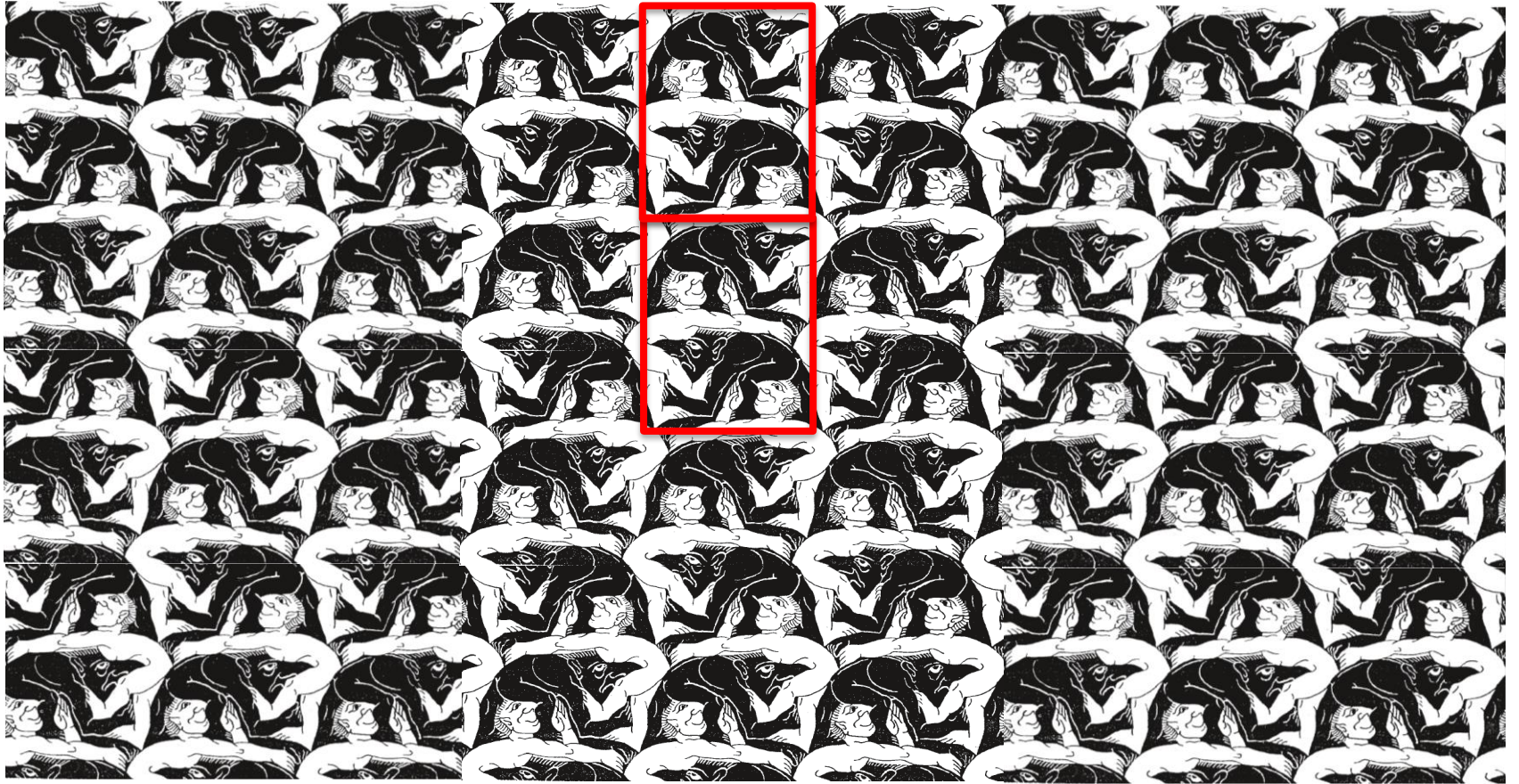
$$\mathbf{L}_{12} = \mathbf{L}_1 + \mathbf{L}_2$$

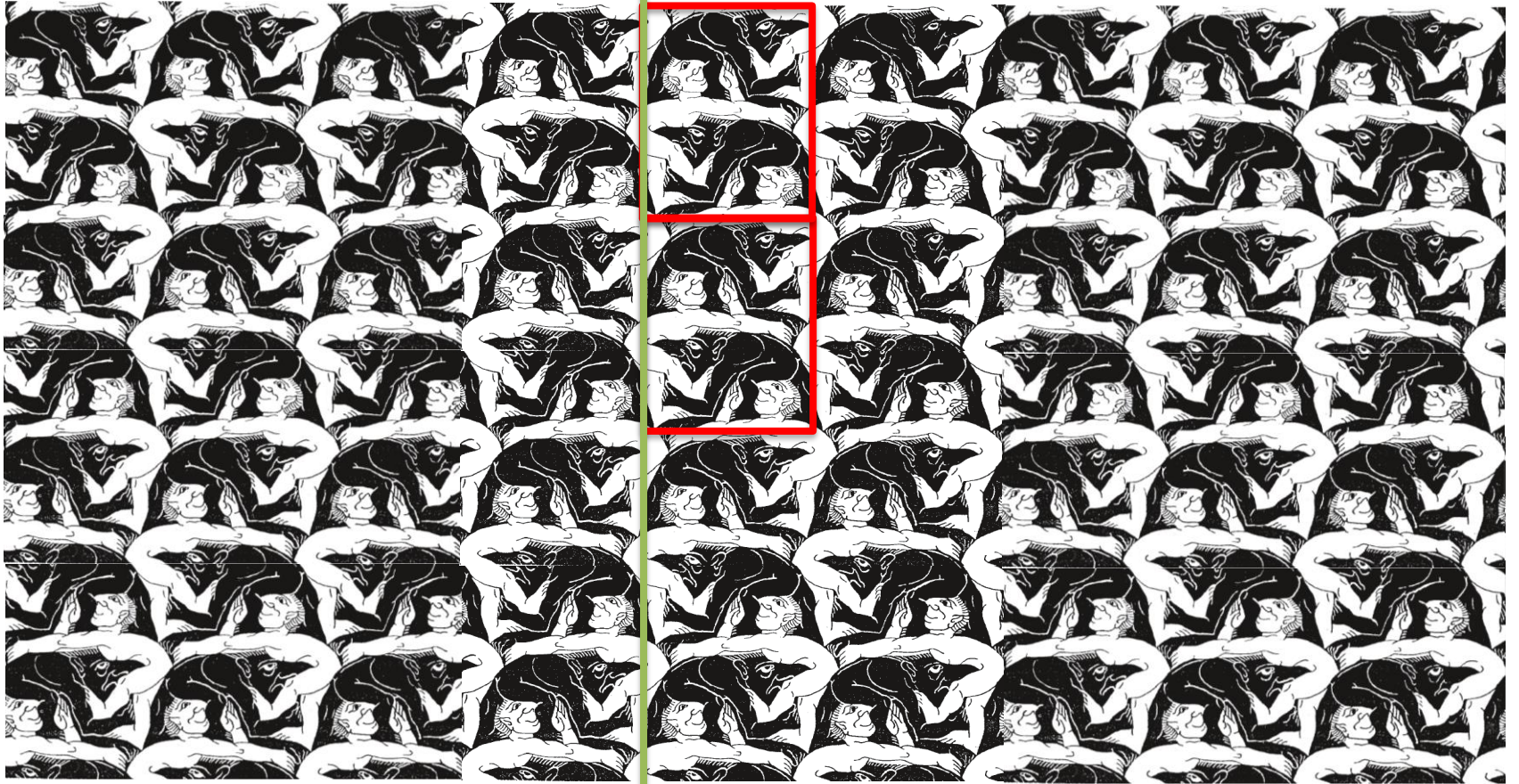
Grupo de simetria espacial:

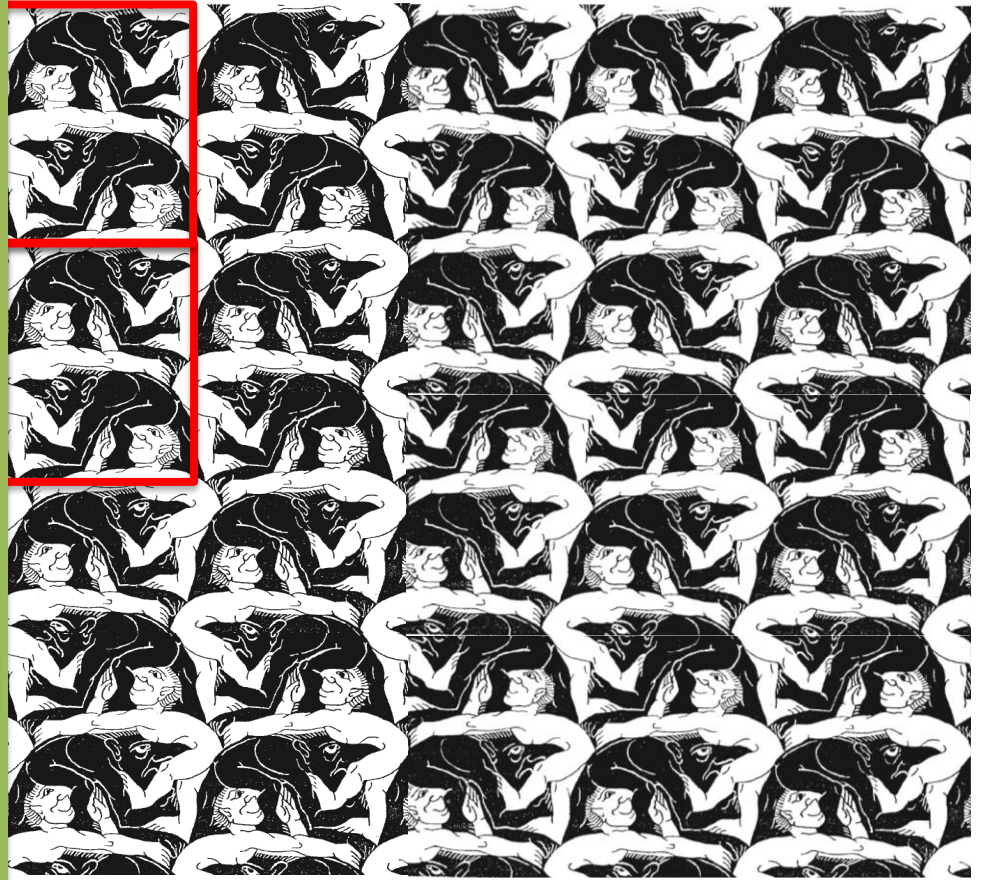
- 1) Translação de todos os pontos da rede por um vetor **R**;
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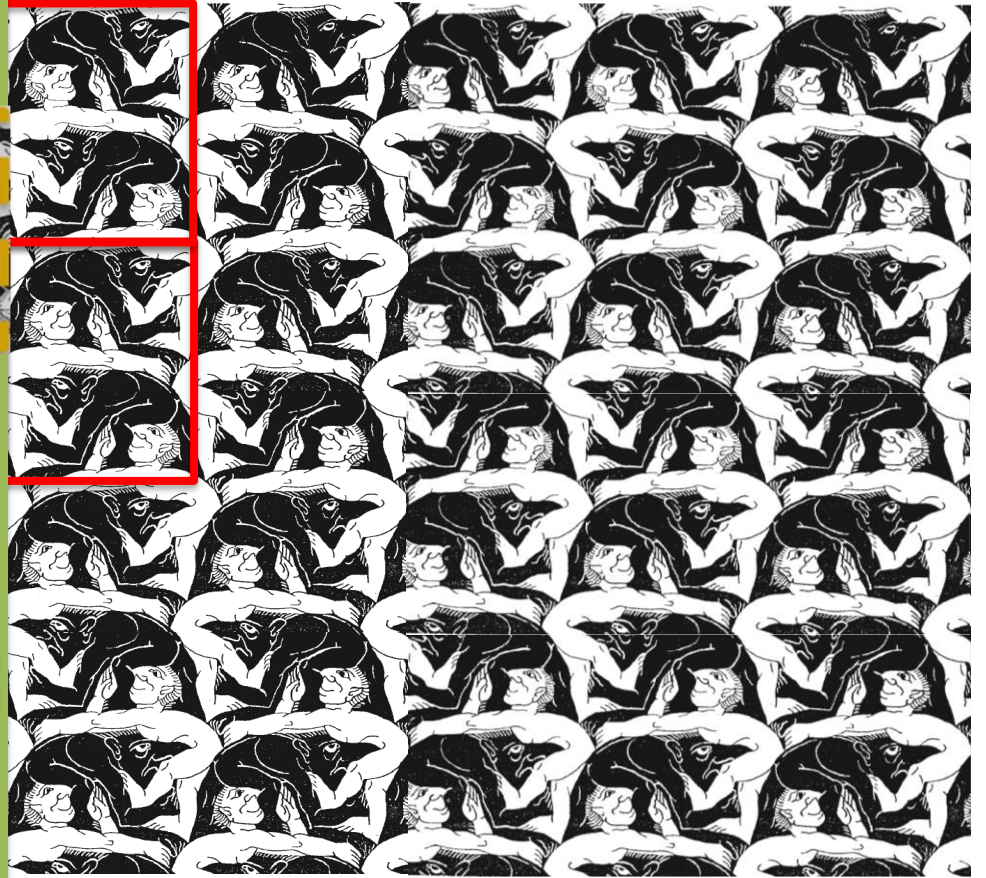
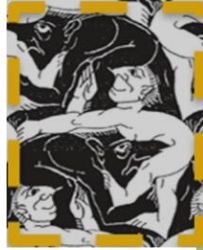


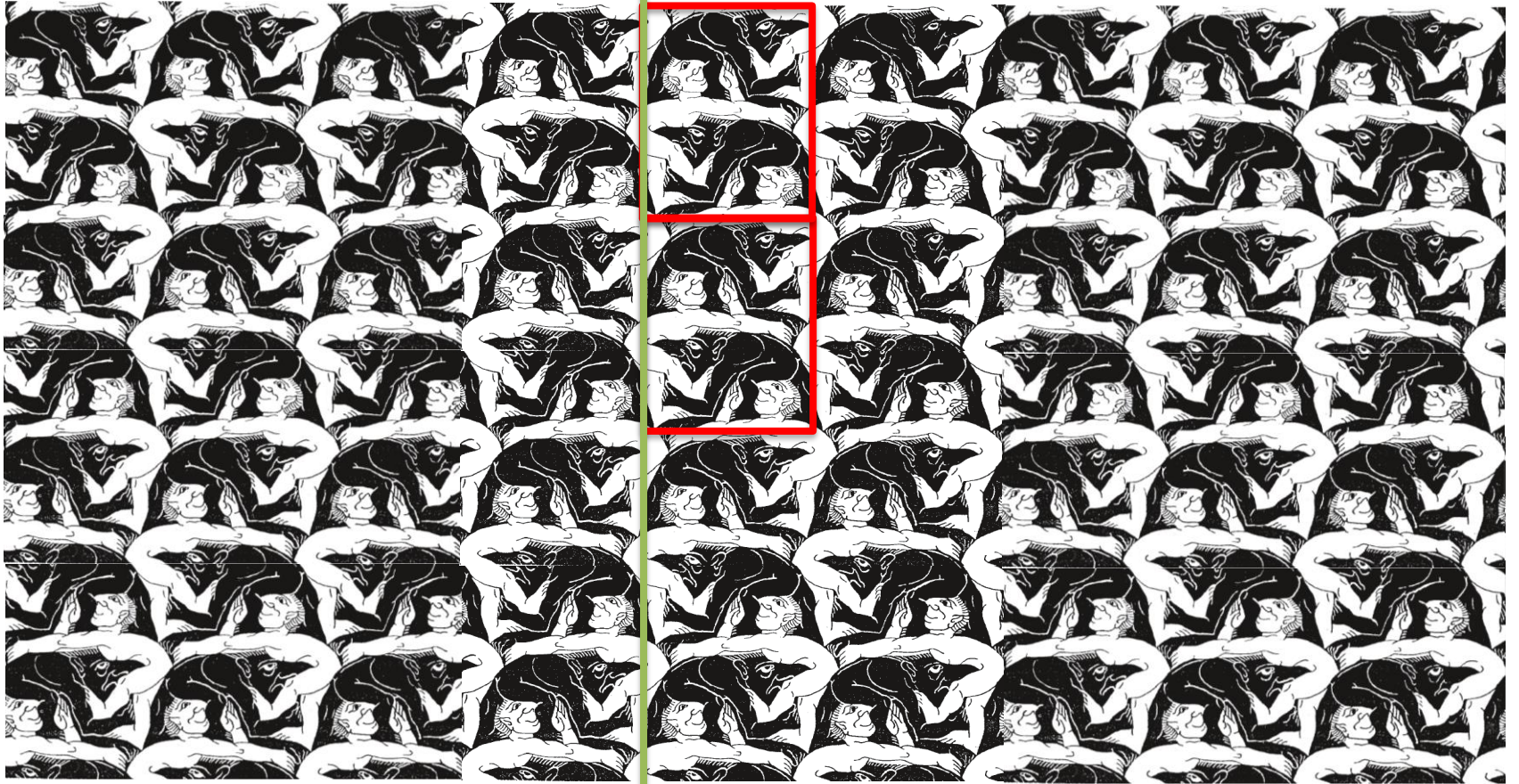












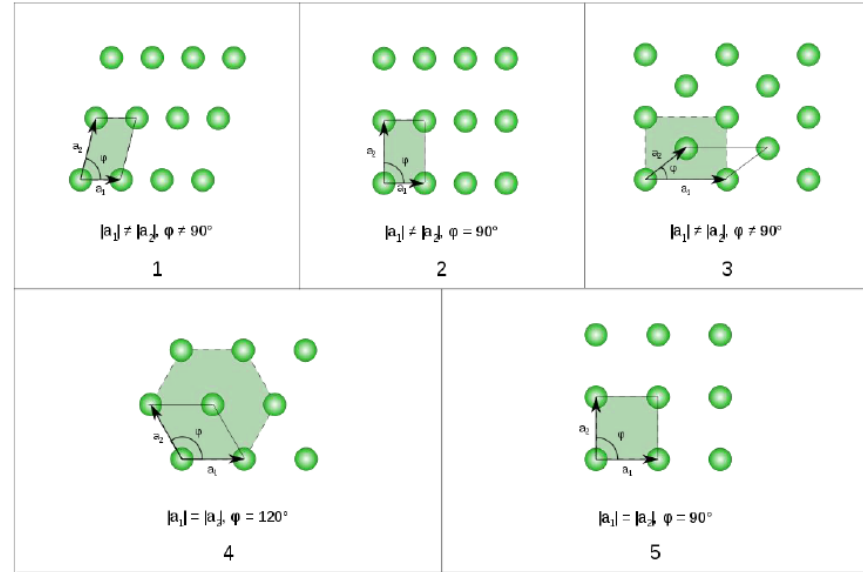
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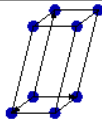
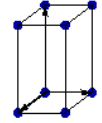
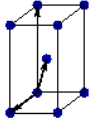
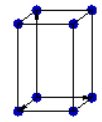
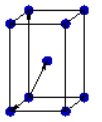
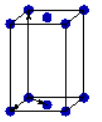
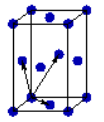
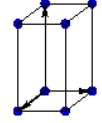
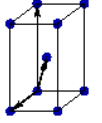
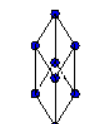
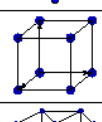
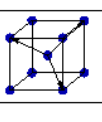
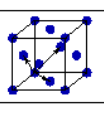
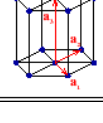
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Em 2D: Temos **cinco** redes de Bravais não-equivalentes!

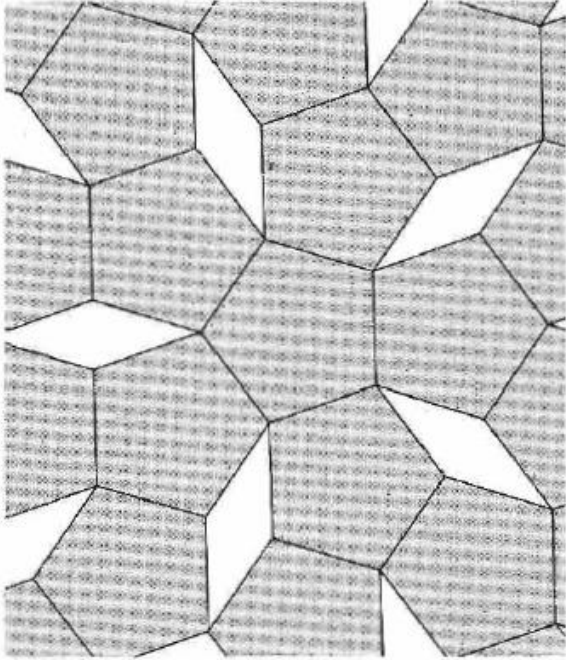


Em 3D: Temos 14 redes de Bravais não-equivalentes!

Bravais lattice	Parameters	Simple (P)	Volume centered (I)	Base centered (C)	Face centered (F)
Triclinic	$a_1 \neq a_2 \neq a_3$ $\alpha_{12} \neq \alpha_{23} \neq \alpha_{31}$				
Monoclinic	$a_1 \neq a_2 \neq a_3$ $\alpha_{23} = \alpha_{31} = 90^\circ$ $\alpha_{12} \neq 90^\circ$				
Orthorhombic	$a_1 \neq a_2 \neq a_3$ $\alpha_{12} = \alpha_{23} = \alpha_{31} = 90^\circ$				
Tetragonal	$a_1 = a_2 \neq a_3$ $\alpha_{12} = \alpha_{23} = \alpha_{31} = 90^\circ$				
Trigonal	$a_1 = a_2 = a_3$ $\alpha_{12} = \alpha_{23} = \alpha_{31} < 120^\circ$				
Cubic	$a_1 = a_2 = a_3$ $\alpha_{12} = \alpha_{23} = \alpha_{31} = 90^\circ$				
Hexagonal	$a_1 = a_2 \neq a_3$ $\alpha_{12} = 120^\circ$ $\alpha_{23} = \alpha_{31} = 90^\circ$				

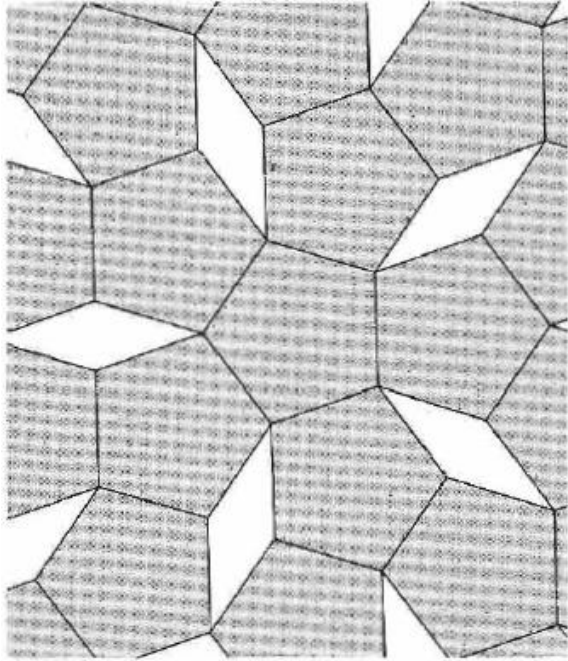
ATENÇÃO: Somente redes com simetrias C_2 , C_3 , C_4 e C_6 são compatíveis com translações em redes 2D e 3D.

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E.g.: A célula unitária de redes com simetria C_5 não preenche todo o espaço!!

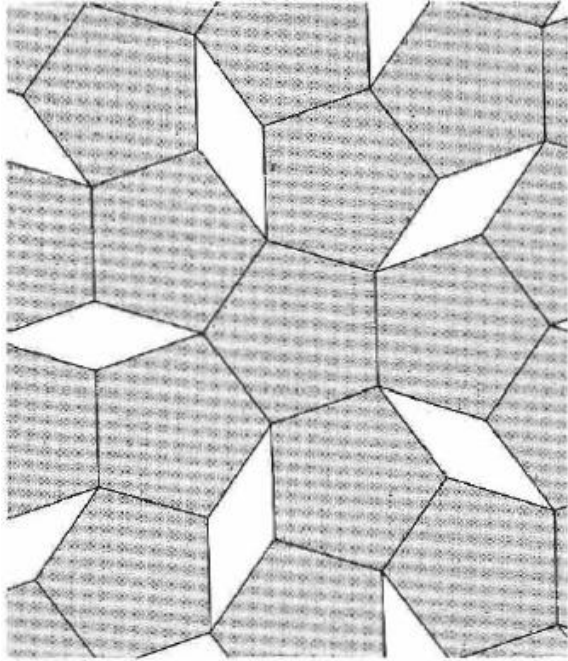
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Demonstração:

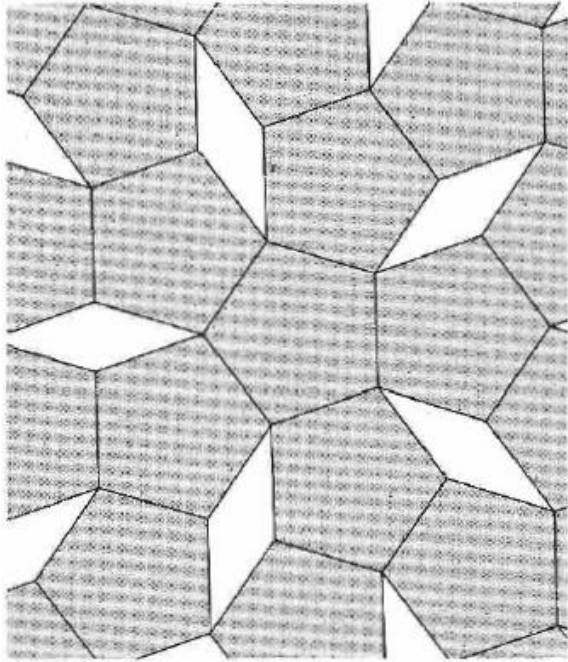
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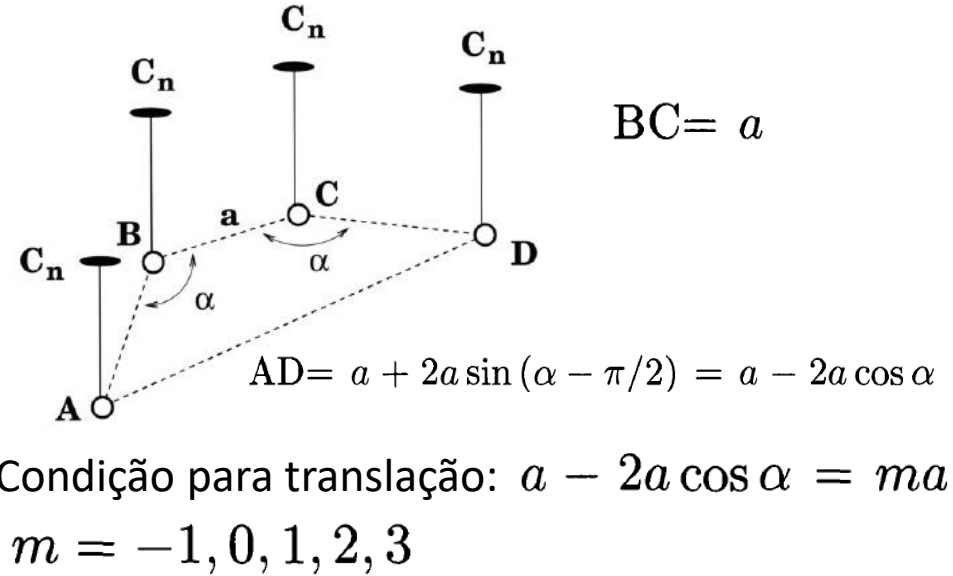
~~Demonstração~~ **mostração:**

ATENÇÃO: Somente redes com simetrias C_2 , C_3 , C_4 e C_6 são compatíveis com translações em redes 2D e 3D.

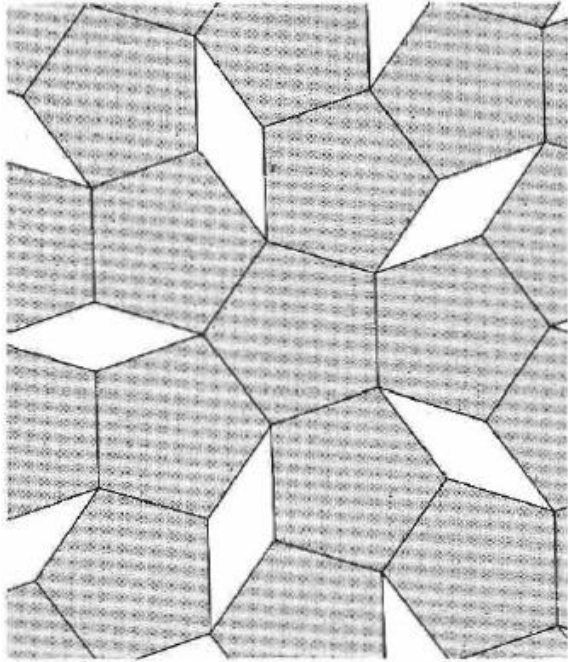


E.g.: A célula unitária de redes com simetria C_5 não preenche todo o espaço!!

~~Demonstração~~ **mostração:**

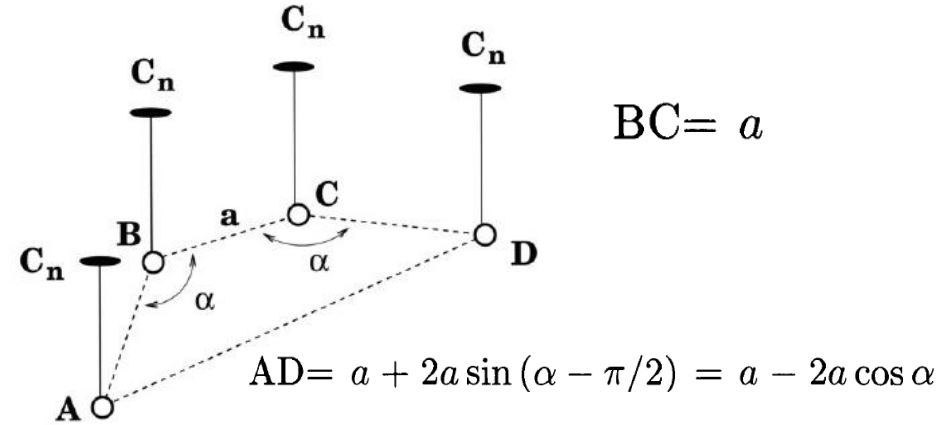


ATENÇÃO: Somente redes com simetrias C_2 , C_3 , C_4 e C_6 são compatíveis com translações em redes 2D e 3D.



E.g.: A célula unitária de redes com simetria C_5 não preenche todo o espaço!!

~~Demonstração~~ **mostração:**



Condição para translação: $a - 2a \cos \alpha = ma$

$$m = -1, 0, 1, 2, 3$$

$$m = -1 \longrightarrow \alpha = 0$$

$$m = 0 \longrightarrow \alpha = 60^\circ = 2\pi/6$$

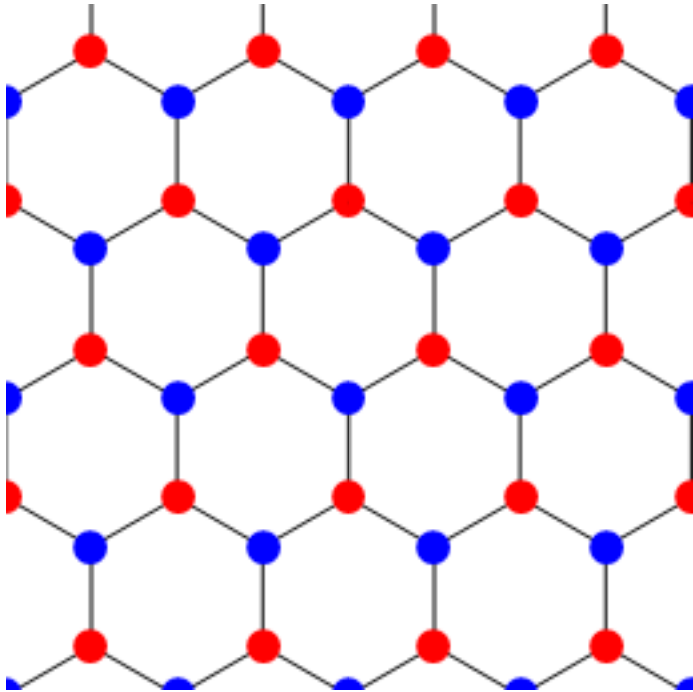
$$m = 1 \longrightarrow \alpha = 90^\circ = 2\pi/4$$

$$m = 2 \longrightarrow \alpha = 120^\circ = 2\pi/3$$

$$m = 3 \longrightarrow \alpha = 180^\circ = 2\pi/2$$

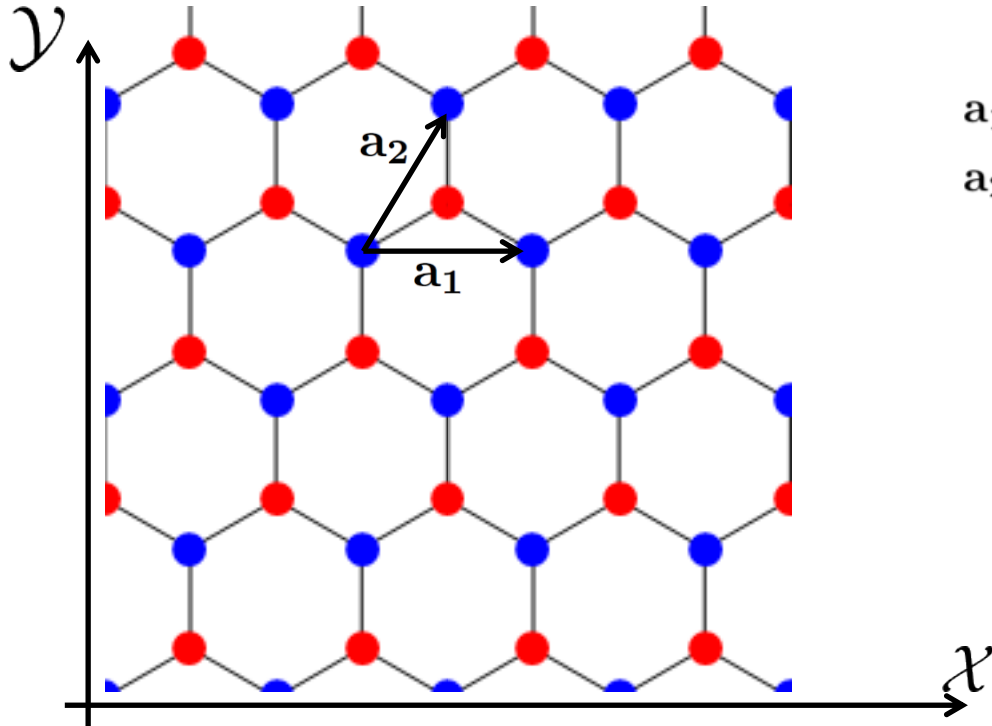
Nem toda rede é uma rede de Bravais

- Ex. Rede Favos de Mel (*Honeycomb*)



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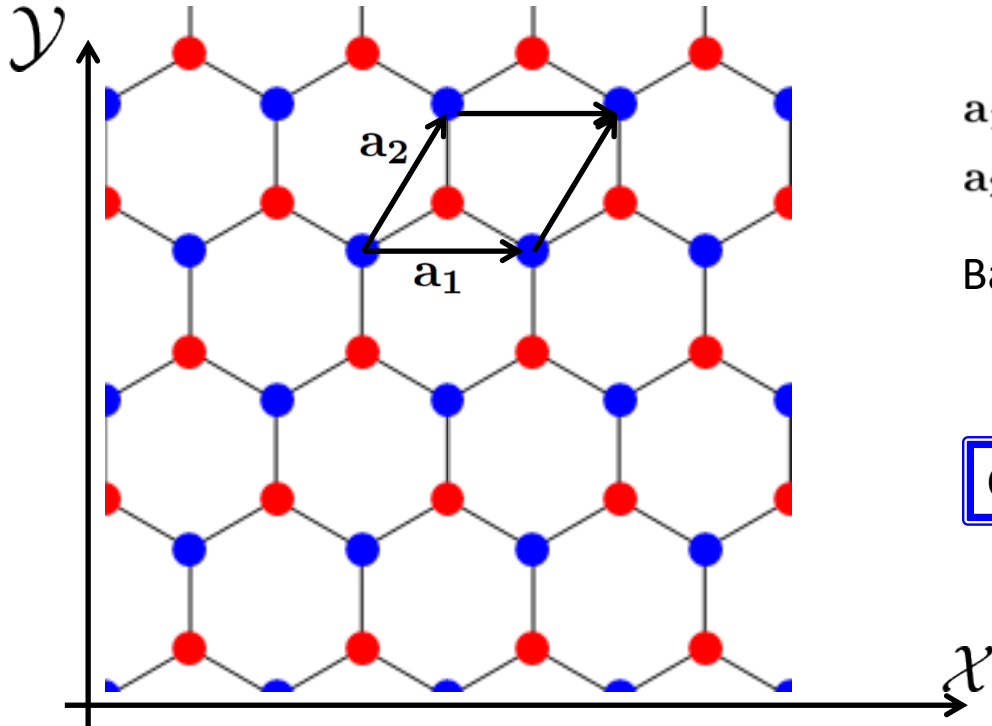


$$\mathbf{a}_1 = a \hat{x}$$

$$\mathbf{a}_2 = (a/2) \hat{x} + (a\sqrt{3}/2) \hat{y}$$

Nem toda rede é uma rede de Bravais

- Ex. Rede Favos de Mel (*Honeycomb*)



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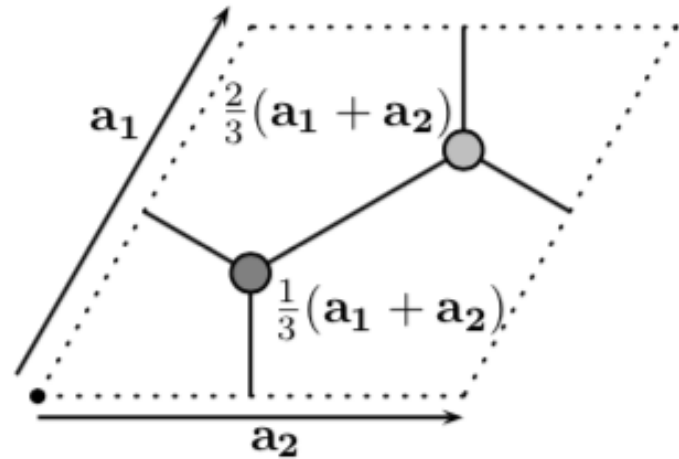
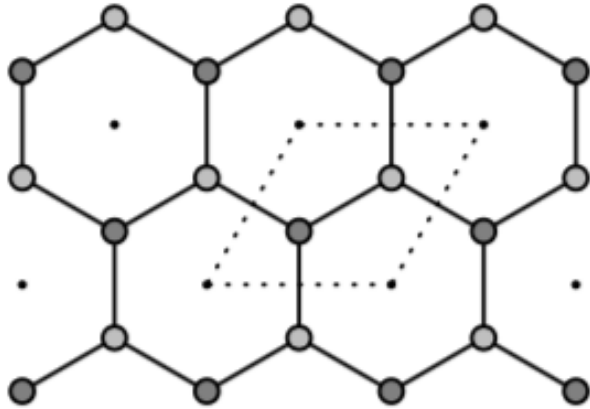
$$\mathbf{a}_2 = (a/2) \hat{x} + (a\sqrt{3}/2) \hat{y}$$

$$\text{Base: } \mathbf{0}, \frac{1}{3}(\mathbf{a}_1 + \mathbf{a}_2)$$

CRISTAL = Rede Hexagonal + Base

Nem toda rede é uma rede de Bravais

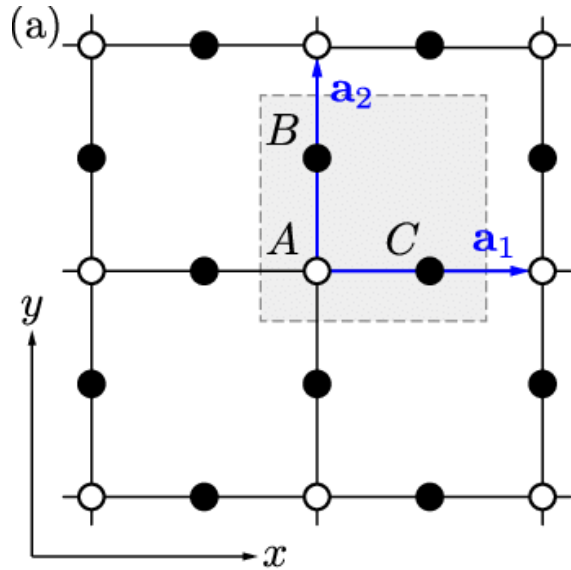
- Ex. Rede Favos de Mel (*Honeycomb*)



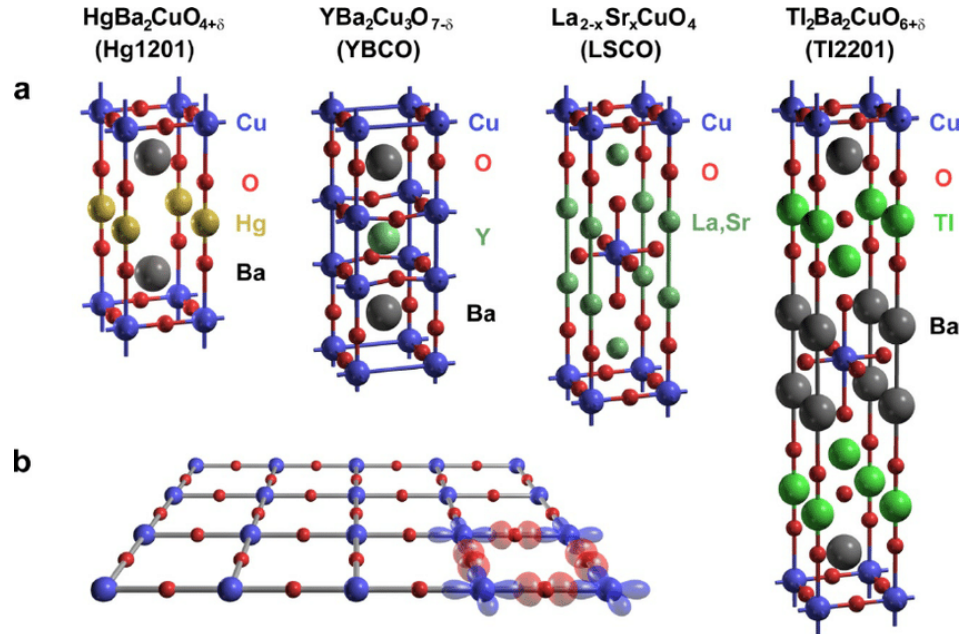
CRISTAL = Rede Hexagonal + Base

Nem toda rede é uma rede de Bravais

- Rede de Lieb.



CRISTAL = Rede quadrada + Base

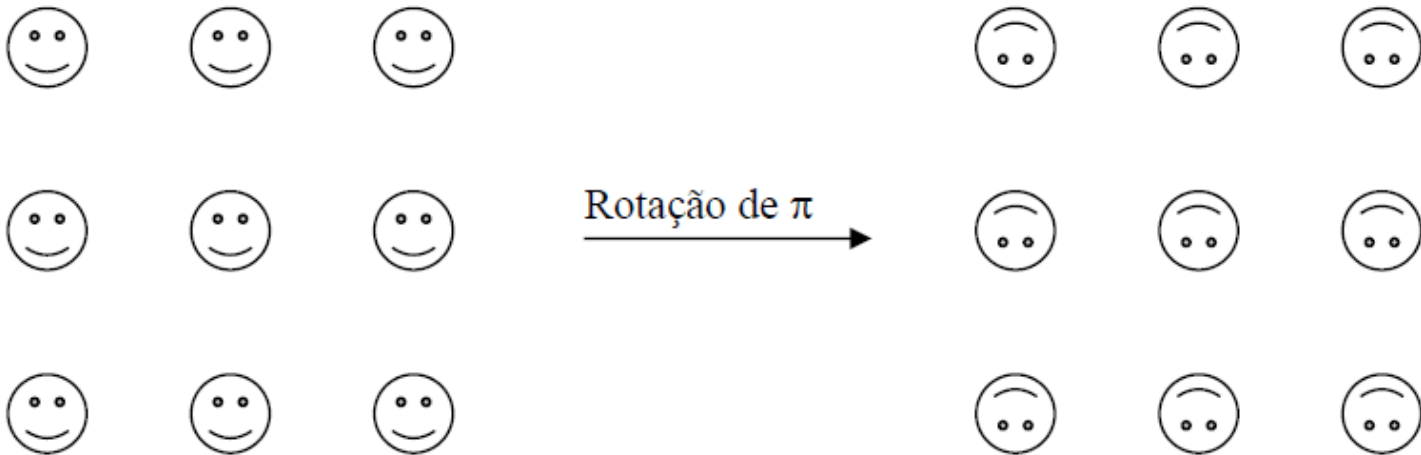


MAIS UMA VEZ... CUIDADO!

A estrutura cristalina pode não conter todas as simetrias que a sua rede de Bravais associada possui!

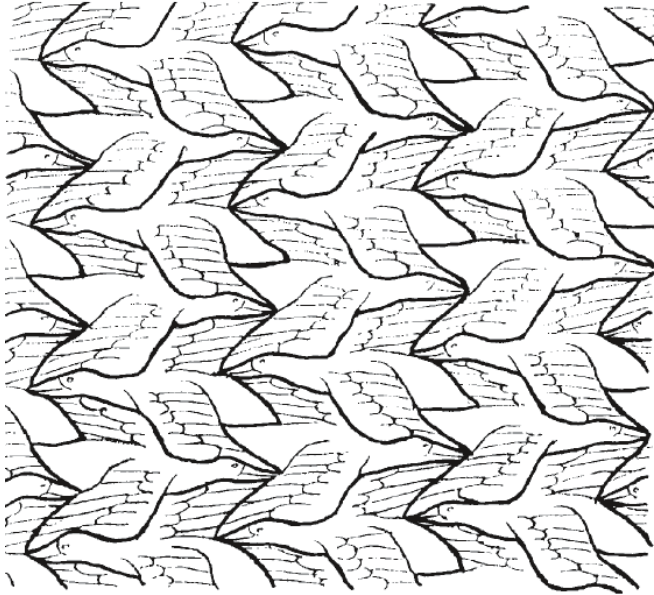
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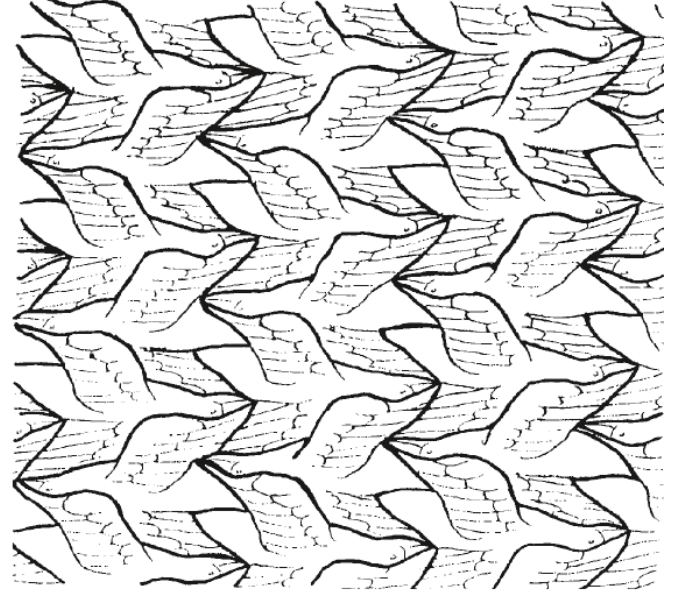


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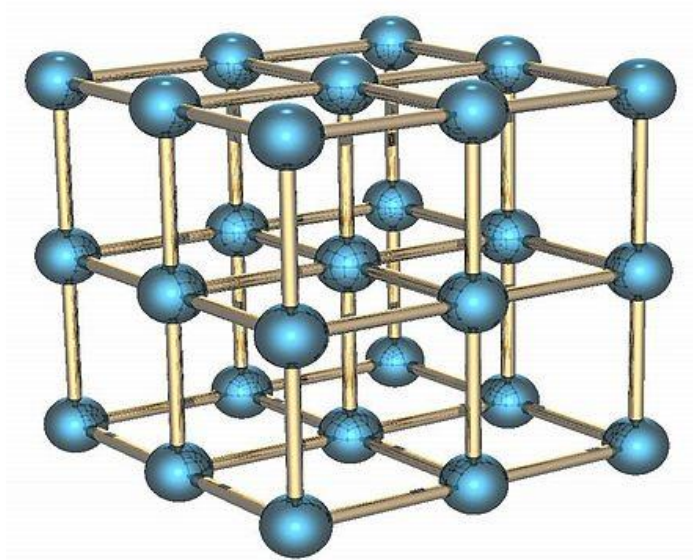


Rotação de π



Redes de Bravais tridimensionais

- Rede cúbica simples Polônio



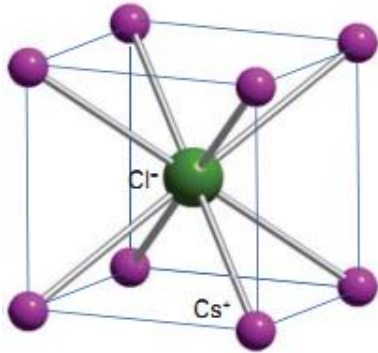
$$a = 3.35 \text{ \AA}$$

94	Po	84
Xe 4f ¹⁴ 5d ¹⁰ 6s ² 6p ⁴		
3.35	SC	
527		

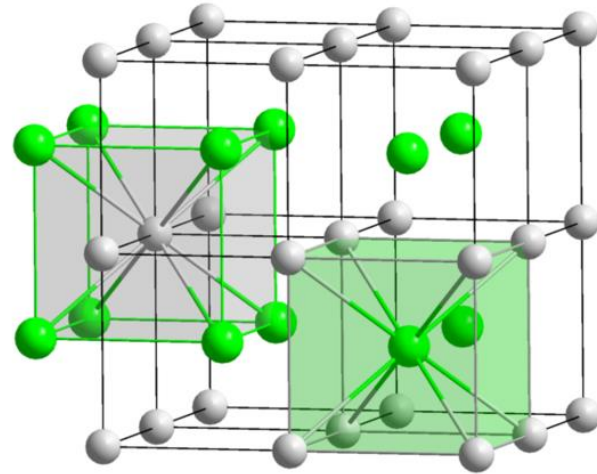
Redes de Bravais tridimensionais

- Rede cúbica simples

Cloreto de Césio

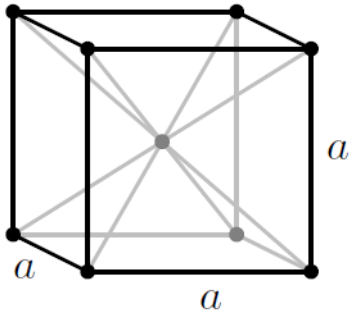


$a = 0.4119 \text{ nm}$

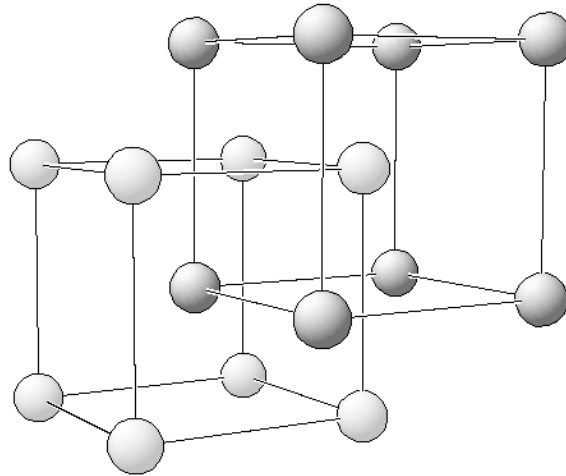


Rede de corpo centrado (BCC)

- rede de corpo centrado (BCC).



Body-centered cubic
unit cell



bcc lattice (2 x sc)

Vetores Primitivos

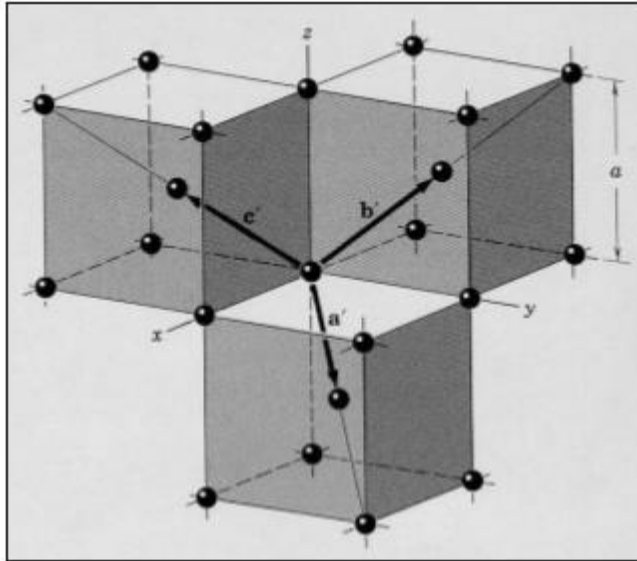
$$\mathbf{a}_1 = [1, 0, 0]$$

$$\mathbf{a}_2 = [0, 1, 0]$$

$$\mathbf{a}_3 = \left[\frac{1}{2}, \frac{1}{2}, \frac{1}{2} \right]$$

Rede de corpo centrado (BCC)

- Uma escolha mais simétrica para os vetores primitivos é



Vetores Primitivos

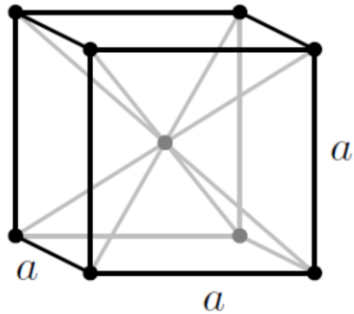
$$\vec{a}' = \frac{a}{2}(\hat{x} + \hat{y} - \hat{z})$$

$$\vec{b}' = \frac{a}{2}(-\hat{x} + \hat{y} + \hat{z})$$

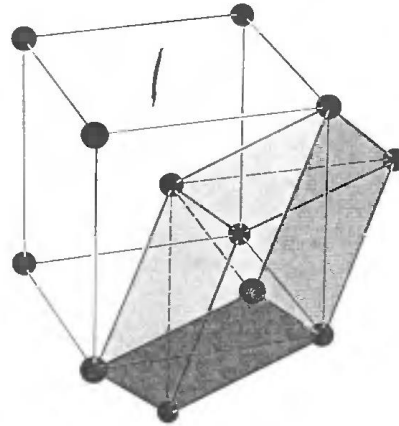
$$\vec{c}' = \frac{a}{2}(\hat{x} - \hat{y} + \hat{z})$$

Rede de corpo centrado (BCC)

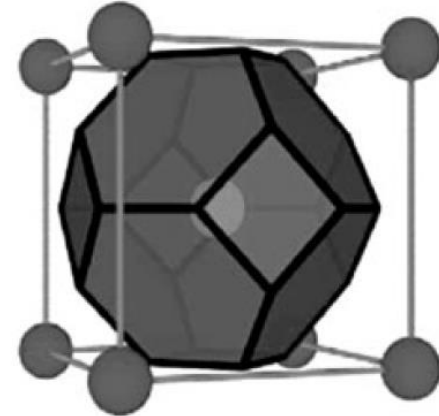
- A célula convencional vs célula primitiva



Célula unitária convencional!



Célula de Wigner-Seitz

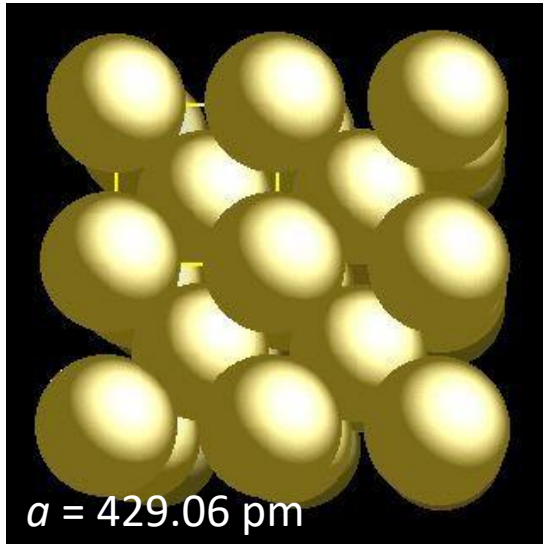


A célula convencional tem o dobro do volume da primitiva!

Octaedro Truncado

Rede de corpo centrado (BCC)

Exemplo: Sódio



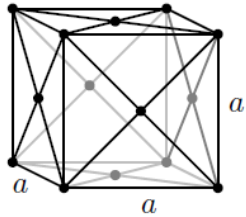
https://www.webelements.com/sodium/crystal_structure_pdb.html

1.53	Rb	37
	[Kr] 5s ¹	
5.59	BCC	
512		56 ^L T
1.90	Cs	55
	[Xe] 6s ¹	
6.05	BCC	
302		40 ^L T
	Fr	87
	[Rn] 7s ¹	
	(BCC)	
3001		

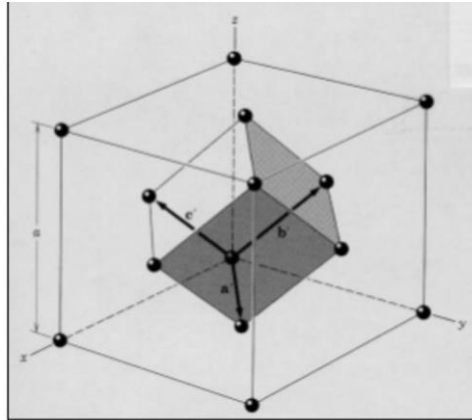
0.53	Li	3
	1s ² 2s ¹	
3.49	BCC	
453		400
0.97	Na	11
	[Ne] 3s ¹	
4.23	BCC	
371.0		150
0.86	K	19
	[Ar] 4s ¹	
5.23	BCC	
337		100

Rede de face centrada (FCC)

- Outro exemplo importante é a rede de face centrada (FCC).



Face-centered cubic
unit cell



A célula convencional tem o
quádruplo do volume da primitiva!

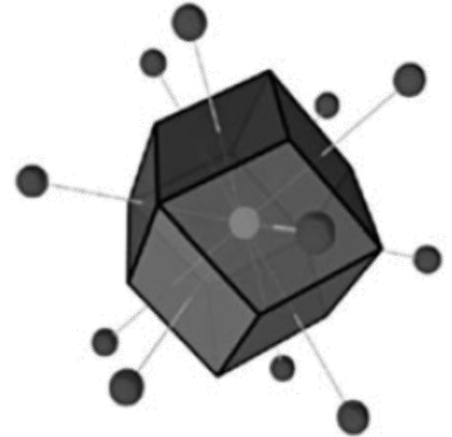
Vetores Primitivos

$$\vec{a}' = \frac{a}{2} (\hat{x} + \hat{y})$$

$$\vec{b}' = \frac{a}{2} (\hat{y} + \hat{z})$$

$$\vec{c}' = \frac{a}{2} (\hat{z} + \hat{x})$$

Célula de Wigner-Seitz

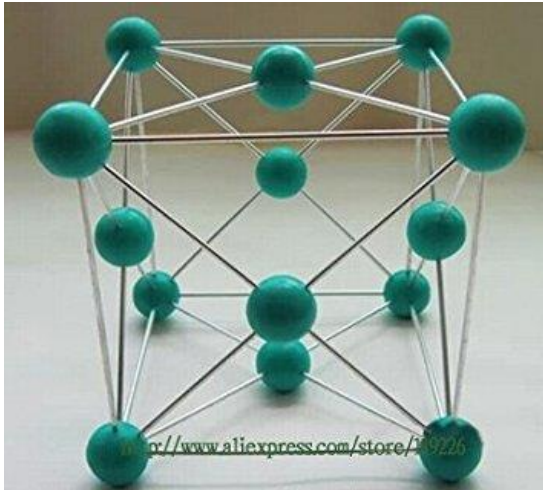


Dodecaedro Rômbico

Rede de face centrada (FCC)

Exemplos: Cobre, Prata e Ouro

$a = 361.49 \text{ pm}$

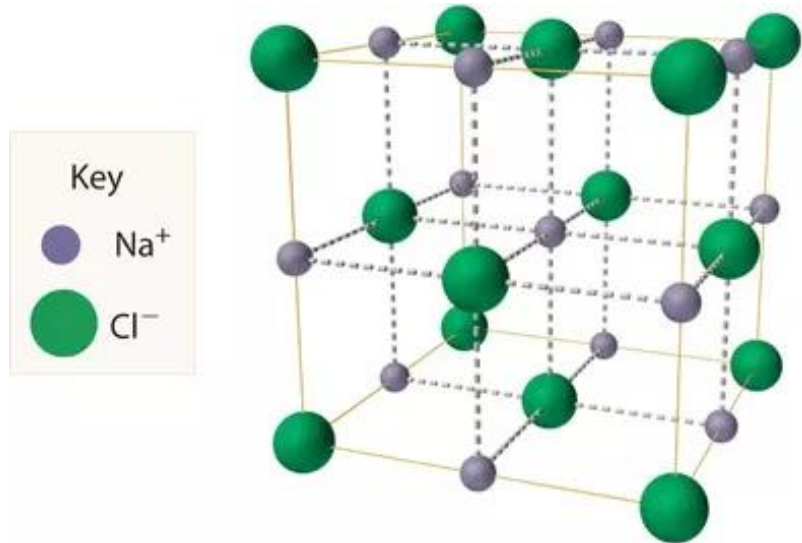


COPPER	63.55	
8.96	Cu	29
	[Ar] 3d ¹⁰ 4s ¹	
3.61	FCC	
1356		315
SILVER	107.87	
10.5	Ag	47
	[Kr] 4d ¹⁰ 5s ¹	
4.09	FCC	
1234		215
GOLD	196.97	
19.3	Au	79
	[Xe] 4f ¹⁴ 5d ¹⁰ 6s ¹	
4.08	FCC	
1337		170

Rede de face centrada (FCC)

Exemplos: Cloreto de Sódio

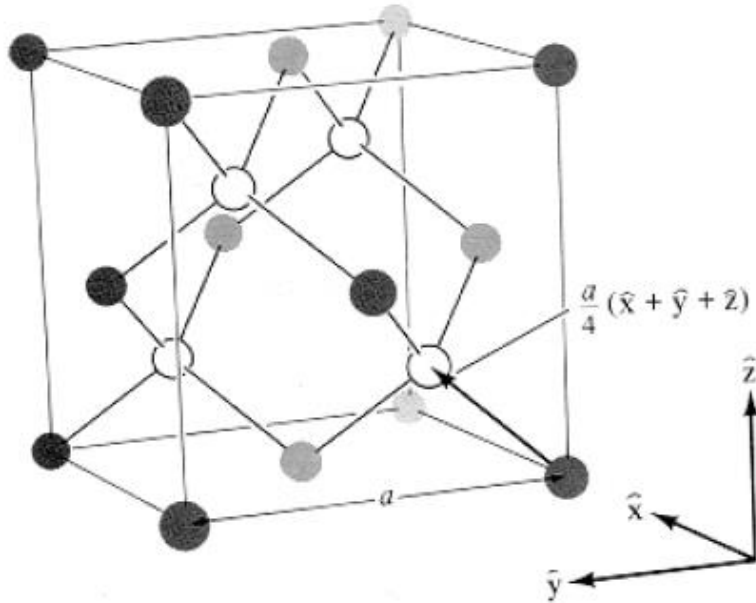
$a = 564.02 \text{ pm}$



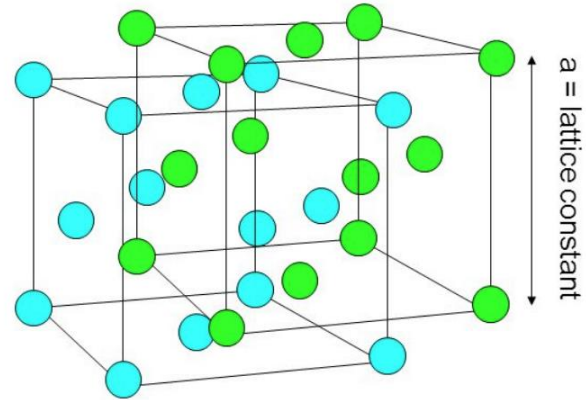
<http://physicsopenlab.org/2018/01/22/sodium-chloride-nacl-crystal/>

Rede de face centrada (FCC)

Exemplos: Diamante



$$a = 3.57 \text{ \AA}$$

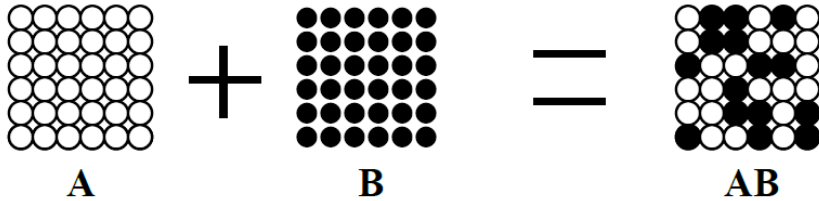


Sólidos desordenados

- Ligas

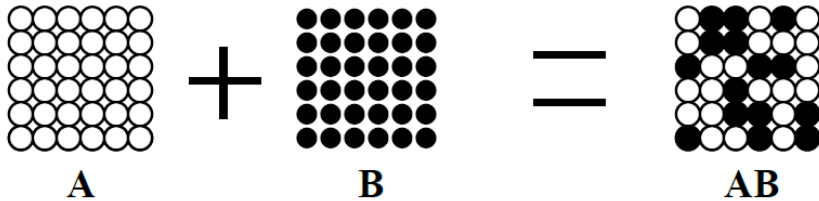
Sólidos desordenados

- Ligas



Sólidos desordenados

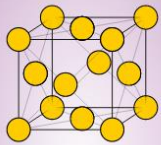
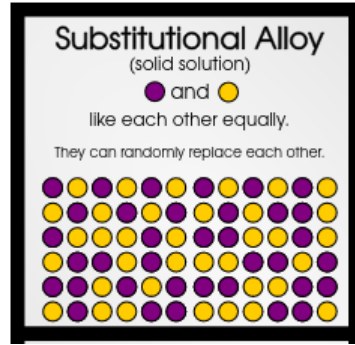
- Ligas



<p>Substitutional Alloy (solid solution) ● and ● like each other equally. They can randomly replace each other.</p>	<p>Interstitial Alloy (solid solution) ● and ● like each other equally. Small atoms randomly squeeze between big atoms.</p>
<p>Intermetallic Compound ● and ● like each other more than themselves They must be arranged in a specific order to maximize contact.</p>	<p>Two-Phase Alloy ● and ● like each other less than themselves They stay in distinct phases to minimize contact</p>

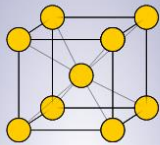
Sólidos desordenados

- Ligas



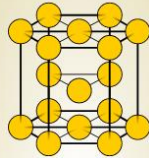
Examples of **FCC**
elements at room
temperature:

Al, Ca, Ni, Cu, Sr, Rh,
Pd, Ag, Yb, Th, Ir, Pt,
Au, Pb



Examples of **BCC**
elements at room
temperature:

Li, Na, K, V, Cr, Mn, Fe,
Rb, Nb, Mo, Cs, Ba, Eu,
Ta, W, Ra

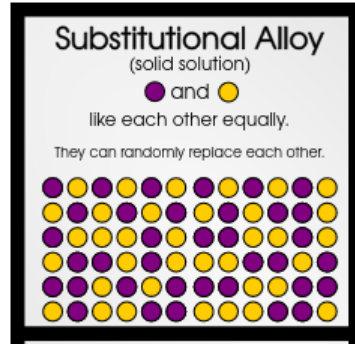


Examples of **HCP**
elements at room
temperature:

Be, Mg, Sc, Ti, Co, Zn,
Y, Zr, Tc, Ru, Cd, Gd,
Tb, Dy, Ho, Er, Tm, Lu,
Hf, Re, Os, Tl

Sólidos desordenados

- Ligas



Copper

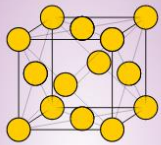
Atomic Number:	29	Atomic Radius:	140 pm (Van der Waals)
Atomic Symbol:	Cu	Melting Point:	1084.6 °C
Atomic Weight:	63.55	Boiling Point:	2562 °C
Electron Configuration:	[Ar]4s ¹ 3d ¹⁰	Oxidation States:	-2, +1, +2 , +3, +4 (a mildly basic oxide)

Silver

Atomic Number:	47	Atomic Radius:	172 pm (Van der Waals)
Atomic Symbol:	Ag	Melting Point:	961.78 °C
Atomic Weight:	107.9	Boiling Point:	2162 °C
Electron Configuration:	[Kr]5s ¹ 4d ¹⁰	Oxidation States:	-2, -1, 1 , 2, 3 (an amphoteric oxide)

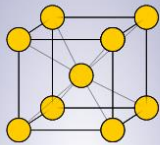
Gold

Atomic Number:	79	Atomic Radius:	166 pm (Van der Waals)
Atomic Symbol:	Au	Melting Point:	1064.18 °C
Atomic Weight:	197.0	Boiling Point:	2970 °C
Electron Configuration:	[Xe]6s ¹ 4f ¹⁴ 5d ¹⁰	Oxidation States:	5 , 3, 2, 1, -1 , -2, -3 (an amphoteric oxide)



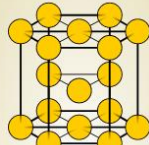
Examples of **FCC** elements at room temperature:

Al, Ca, **Ni**, **Cu**, Sr, Rh,
Pd, **Ag**, Yb, Th, Ir, Pt,
Au, Pb



Examples of **BCC** elements at room temperature:

Li, Na, K, V, **Cr**, Mn, **Fe**,
Rb, **Nb**, **Mo**, Cs, Ba, Eu,
Ta, W, Ra

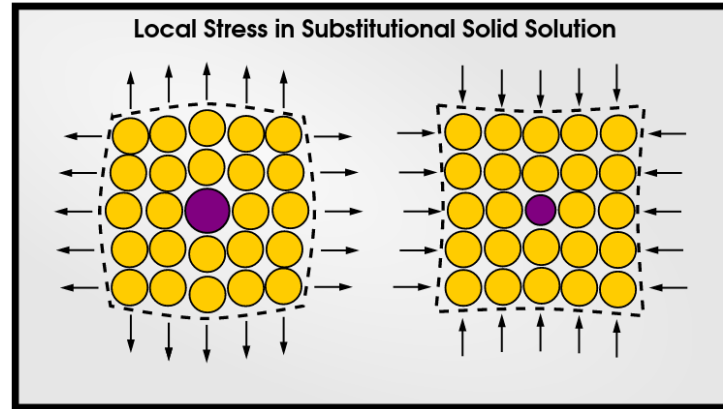


Examples of **HCP** elements at room temperature:

Be, **Mg**, Sc, Ti, **Co**, **Zn**,
Y, **Zr**, Tc, Ru, Cd, Gd,
Tb, Dy, Ho, Er, Tm, Lu,
Hf, Re, Os, Tl

Sólidos desordenados

- Ligas

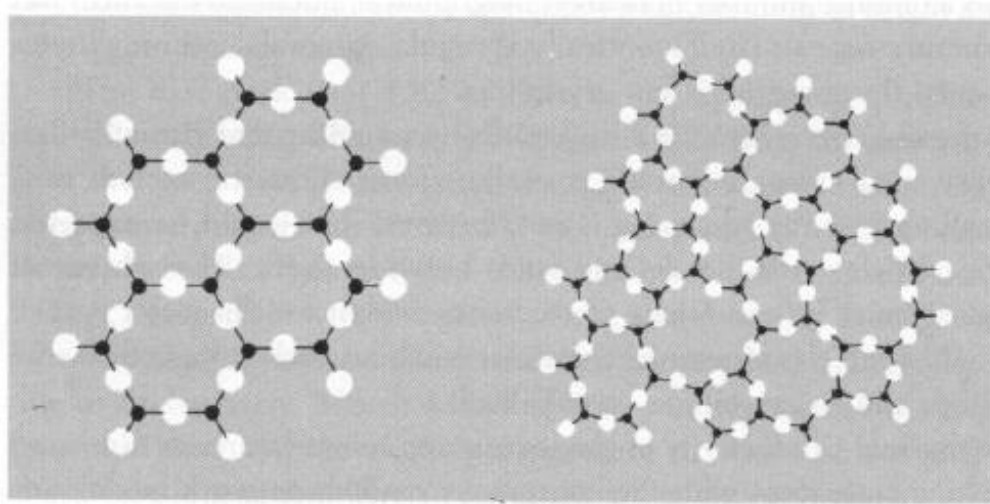


$$\text{Mismatch} = \left(\frac{r_{\text{solute}} - r_{\text{solvent}}}{r_{\text{solvent}}} \right) \times 100 \leq 15\%$$

Hume-Rothary rules

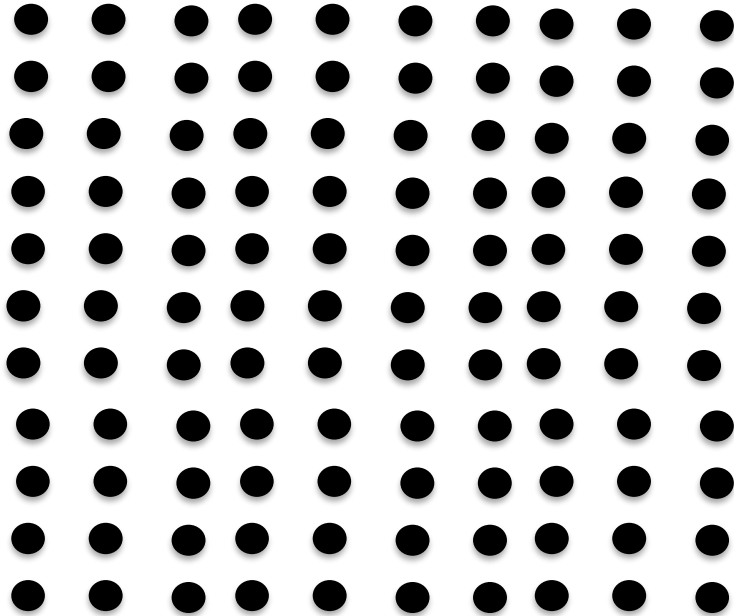
Sólidos desordenados

- Sólidos amorfos



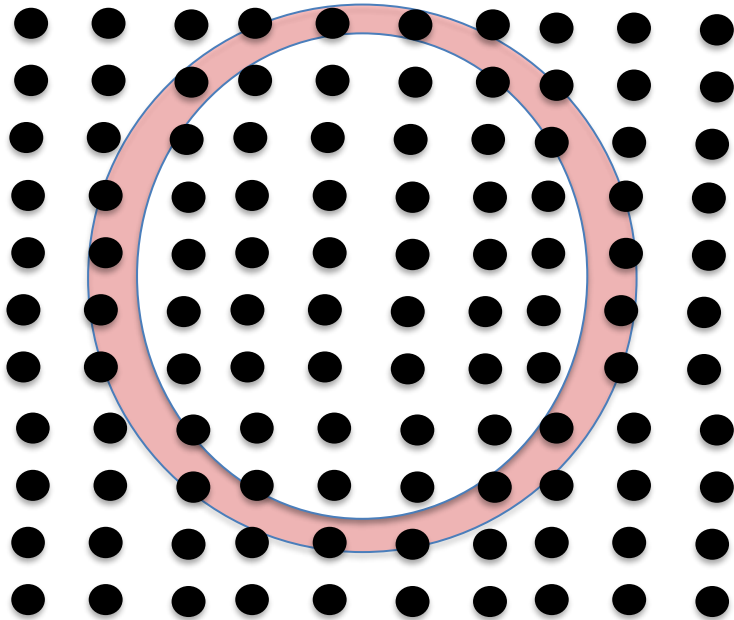
Sólidos desordenados

- Sólidos amorfos



Sólidos desordenados

- Sólidos amorfos



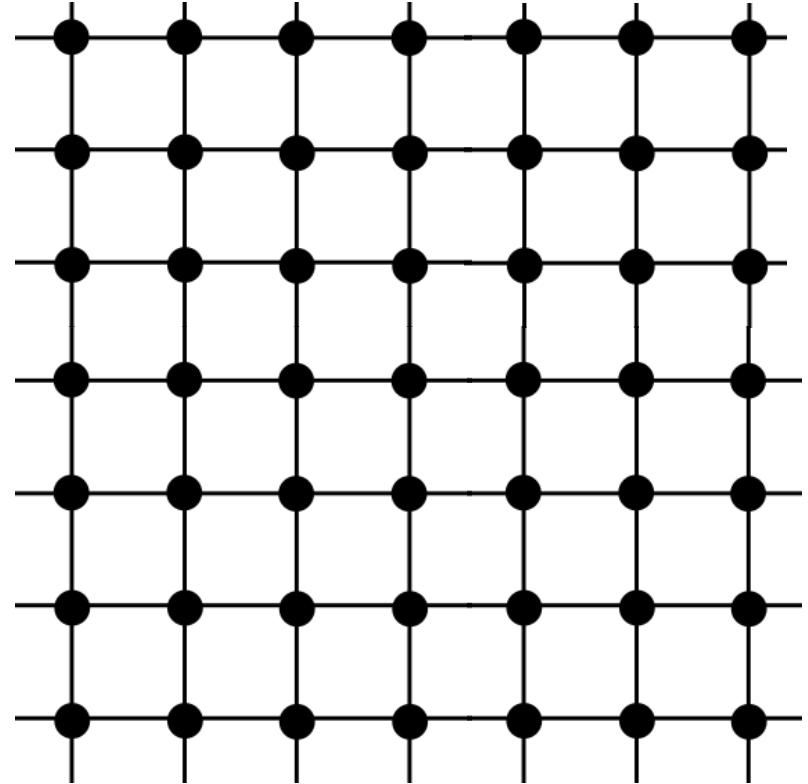
$$g(r) = 4\pi r^2 \rho(r)$$

Sólidos desordenados

- Sólidos amorfos



$$g(r) = 4\pi r^2 \rho(r)$$

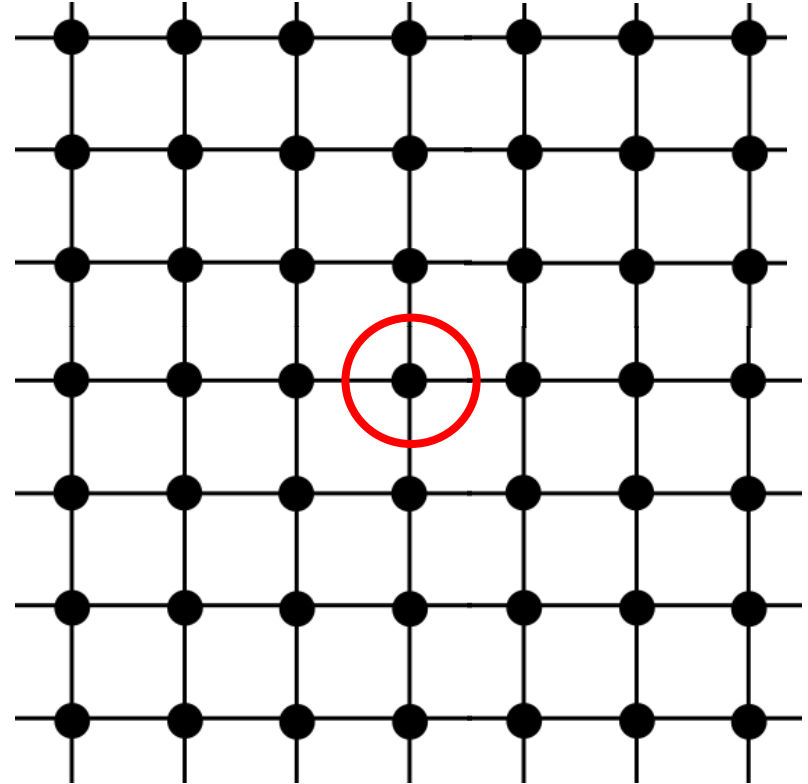


Sólidos desordenados

- Sólidos amorfos

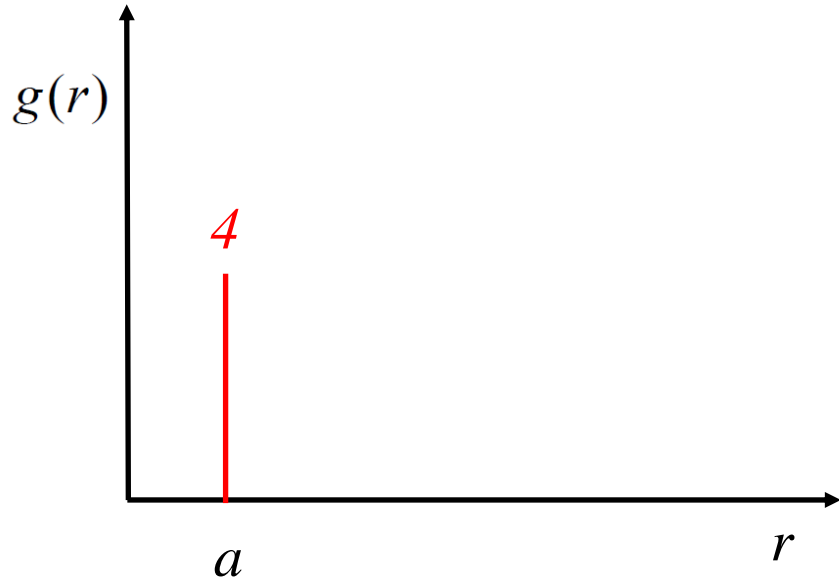


$$g(r) = 4\pi r^2 \rho(r)$$

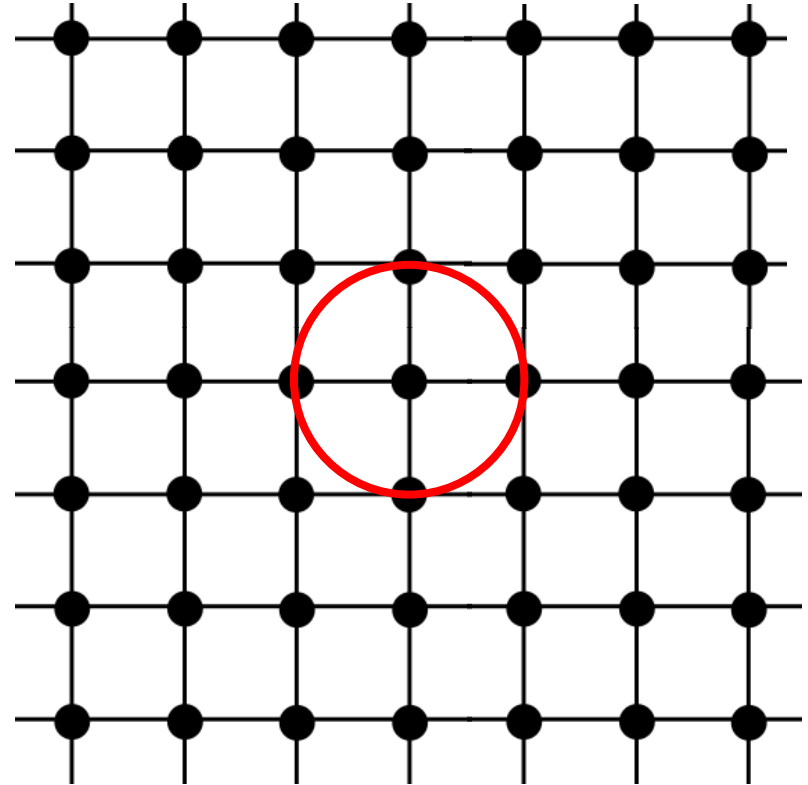


Sólidos desordenados

- Sólidos amorfos

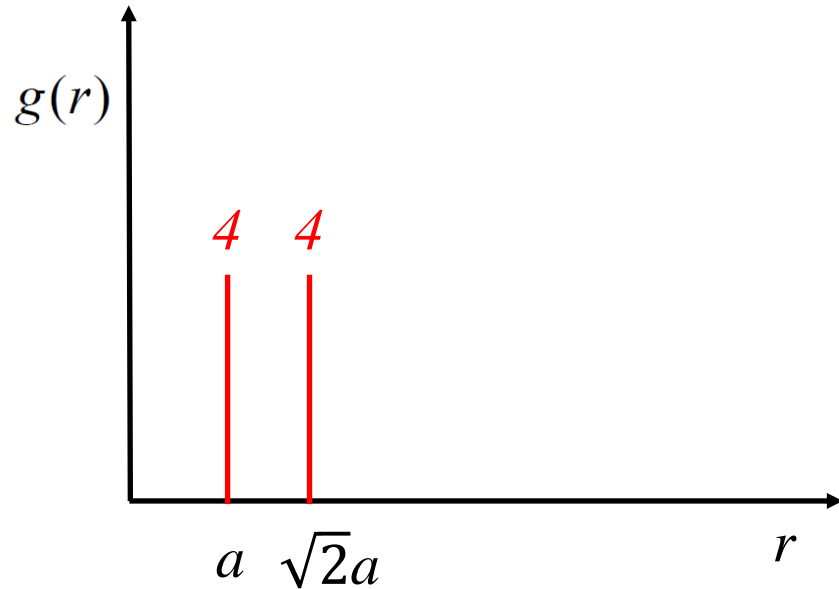


$$g(r) = 4\pi r^2 \rho(r)$$

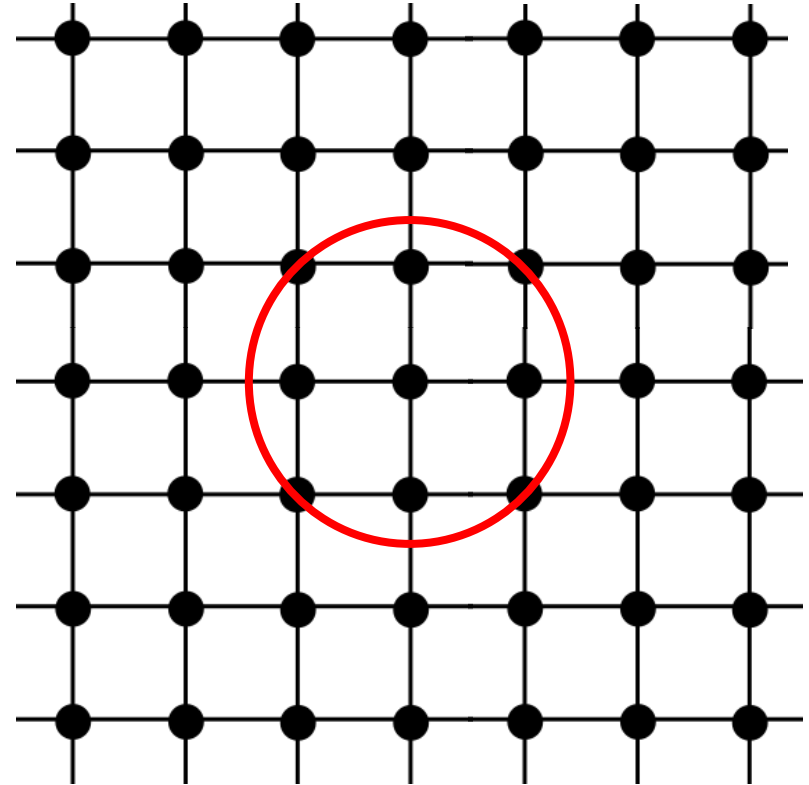


Sólidos desordenados

- Sólidos amorfos

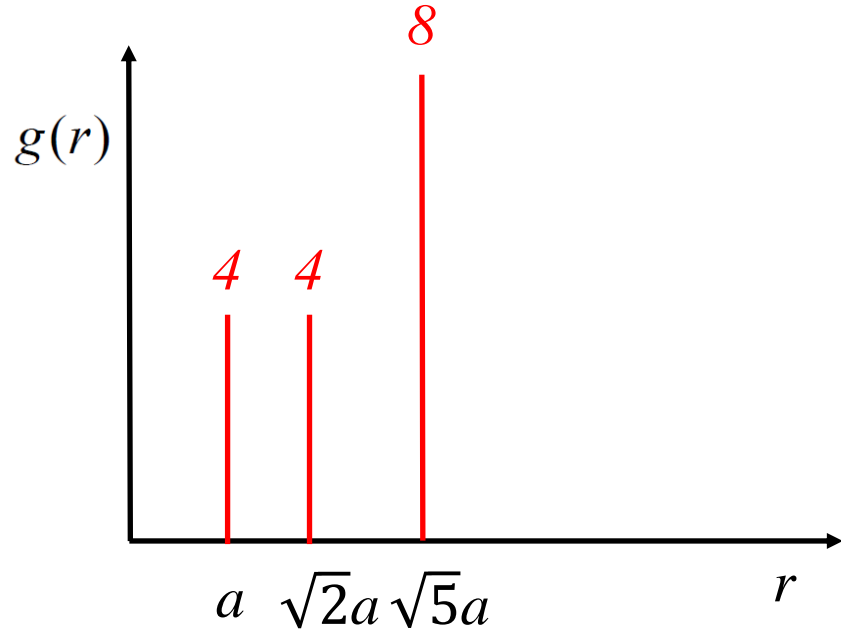


$$g(r) = 4\pi r^2 \rho(r)$$

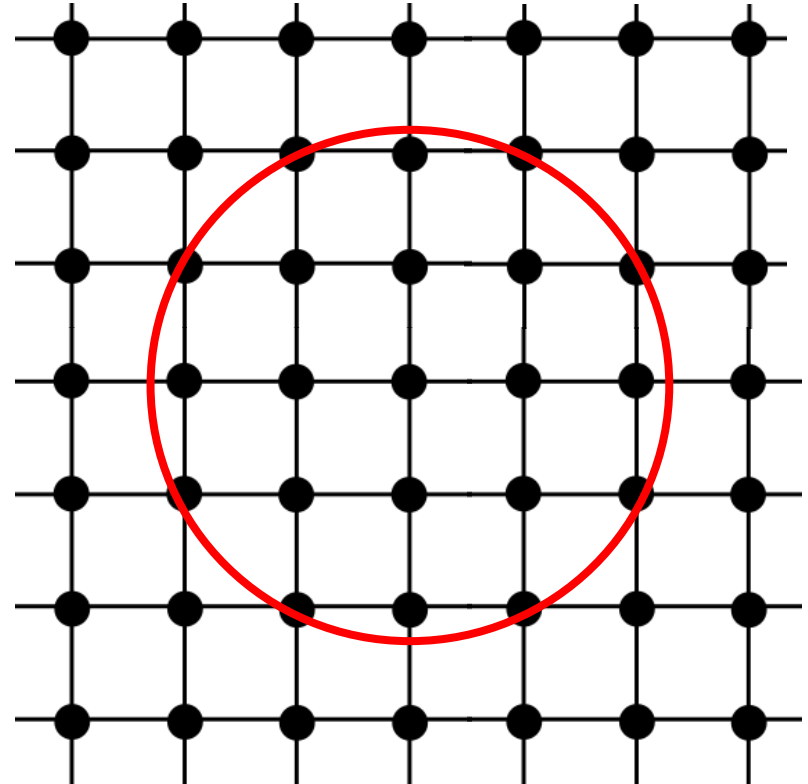


Sólidos desordenados

- Sólidos amorfos

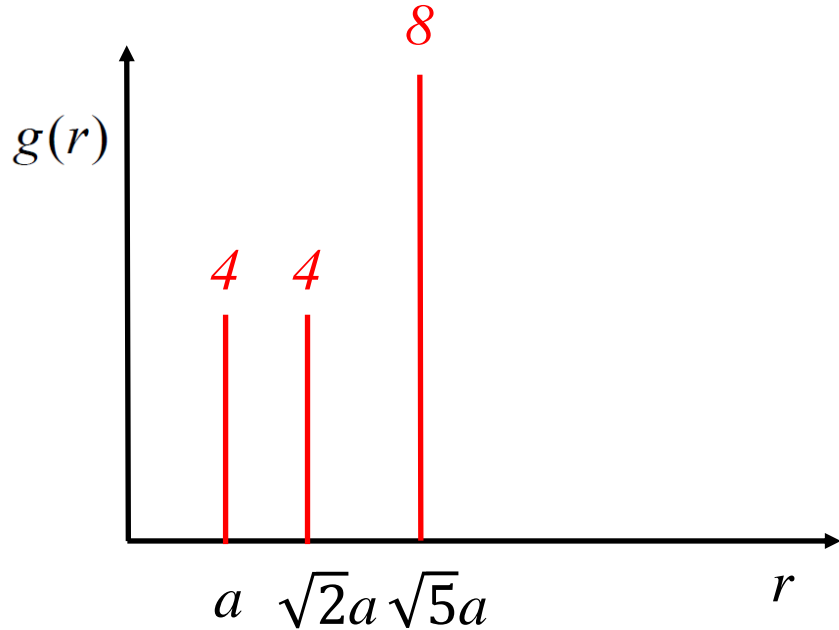


$$g(r) = 4\pi r^2 \rho(r)$$

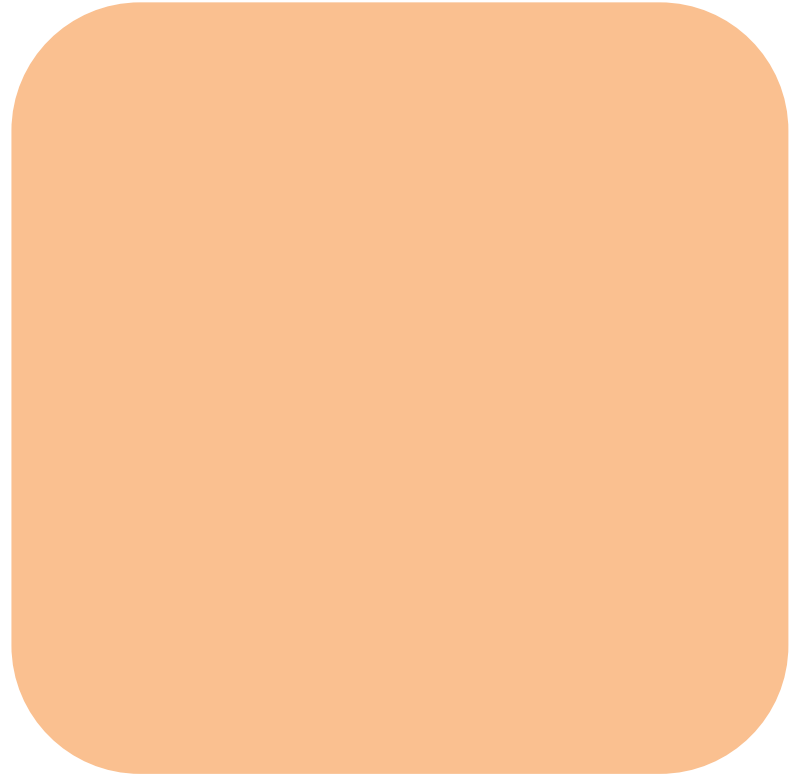


Sólidos desordenados

- Sólidos amorfos

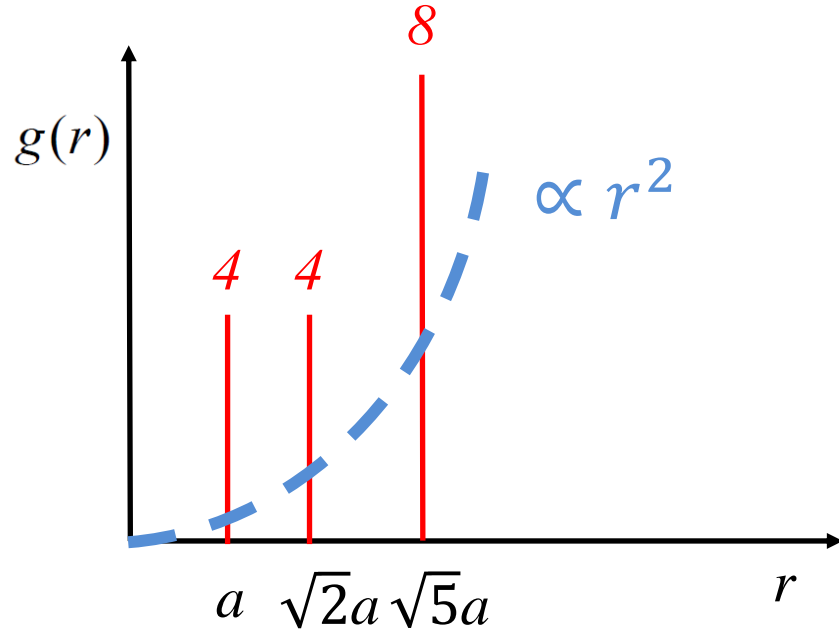


$$g(r) = 4\pi r^2 \rho(r)$$

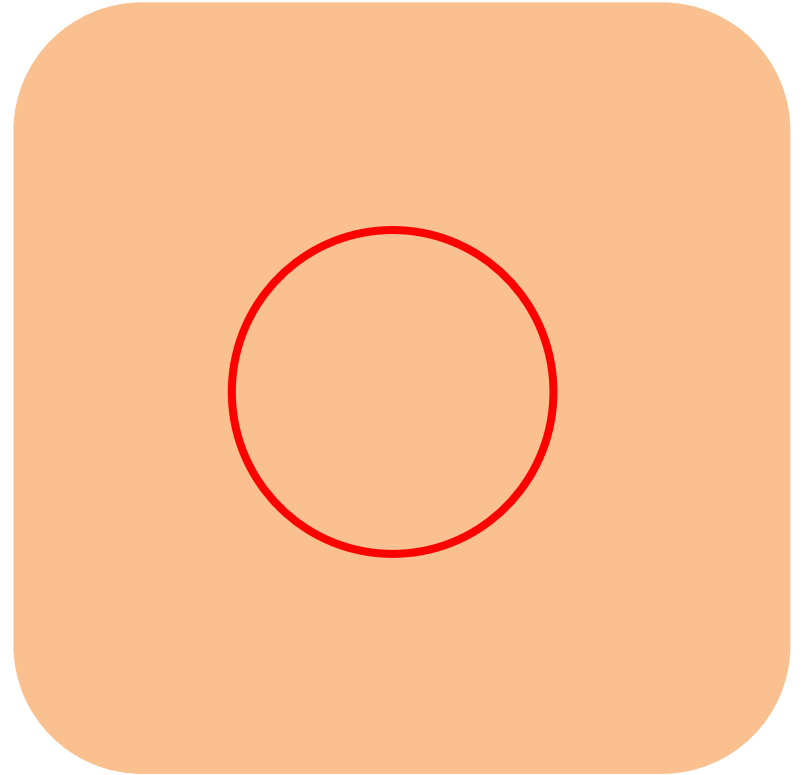


Sólidos desordenados

- Sólidos amorfos



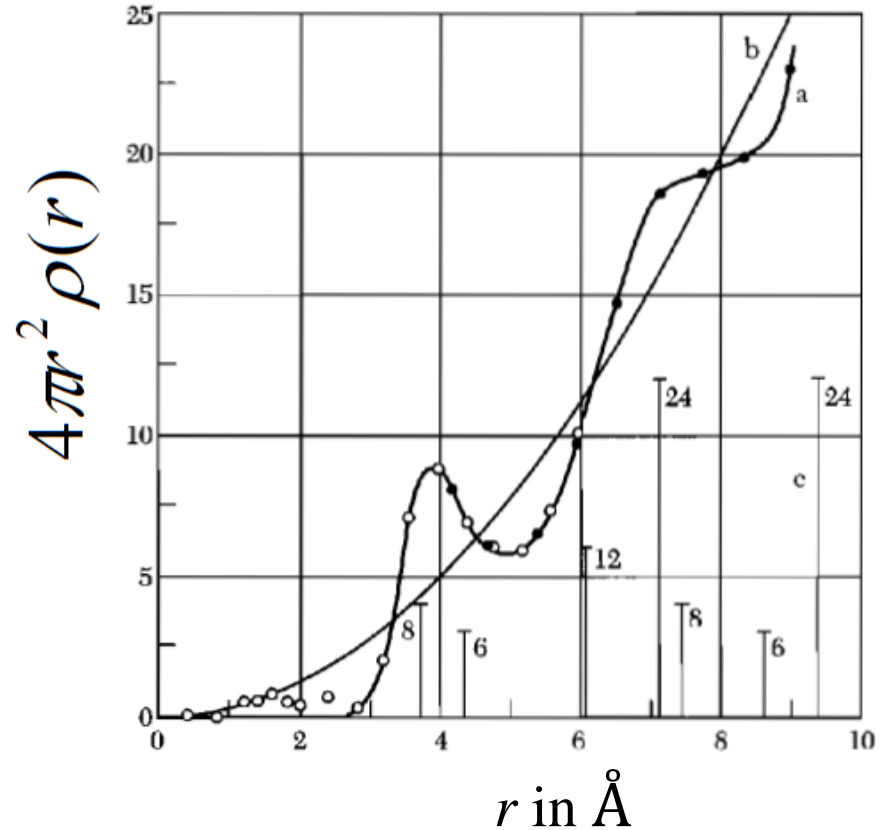
$$g(r) = 4\pi r^2 \rho(r)$$



Sólidos desordenados

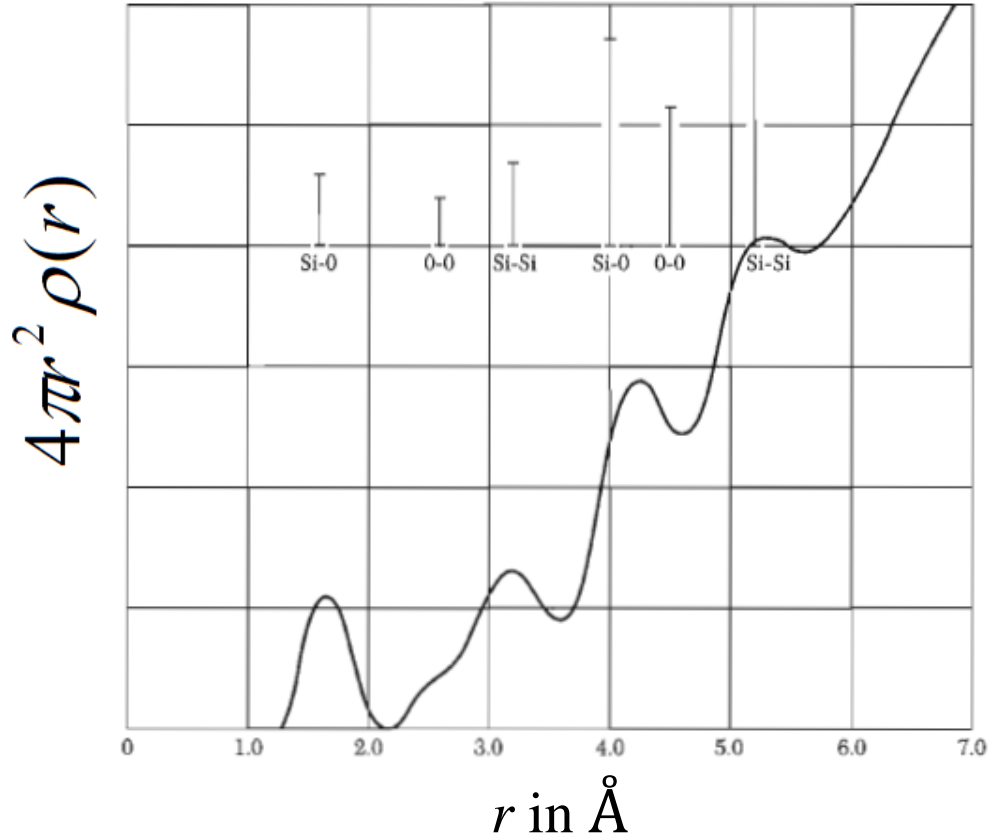
- Sólidos amorfos

Na: sólido, líquido e gasoso.



Sólidos desordenados

- Sólidos amorfos



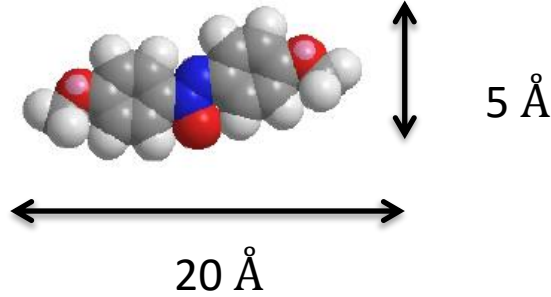
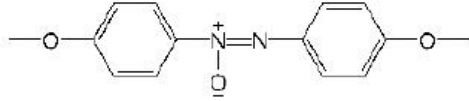
Cristais líquidos

Cristais líquidos

- Crystalline properties: anisotropy of optical, electrical, and magnetic properties, as well as a periodic arrangement of molecules.
- Liquid properties: fluidity and the inability to support shear, formation and coalescence of droplets.

Cristais líquidos

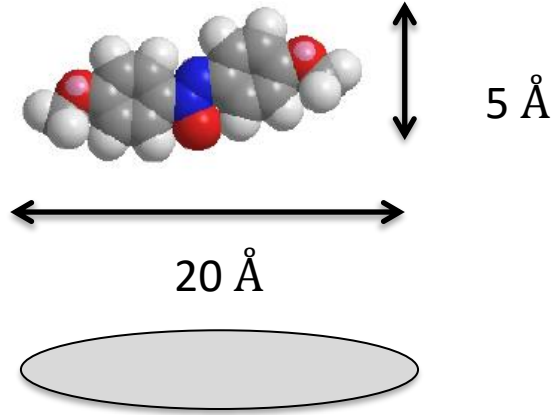
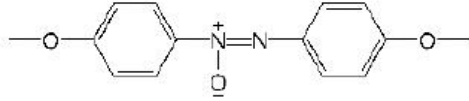
PAA



- Crystalline properties: anisotropy of optical, electrical, and magnetic properties, as well as a periodic arrangement of molecules.
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Cristais líquidos

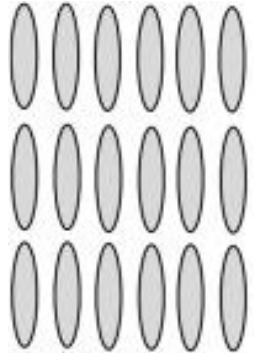
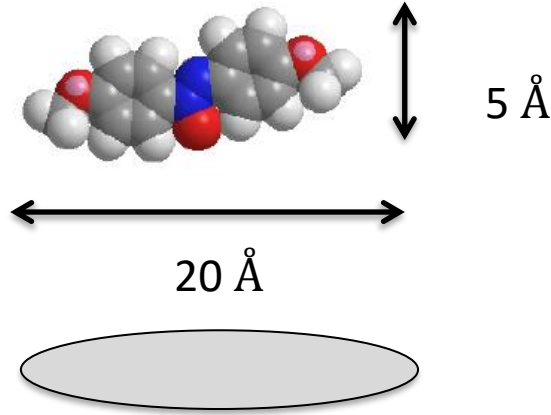
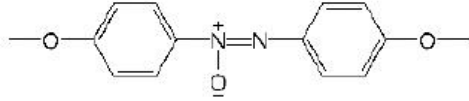
PAA



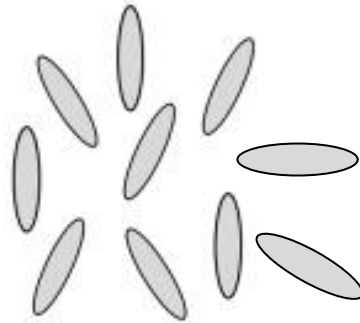
- Crystalline properties: anisotropy of optical, electrical, and magnetic properties, as well as a periodic arrangement of molecules.
- Liquid properties: fluidity and the inability to support shear, formation and coalescence of droplets.

Cristais líquidos

PAA



Crystal

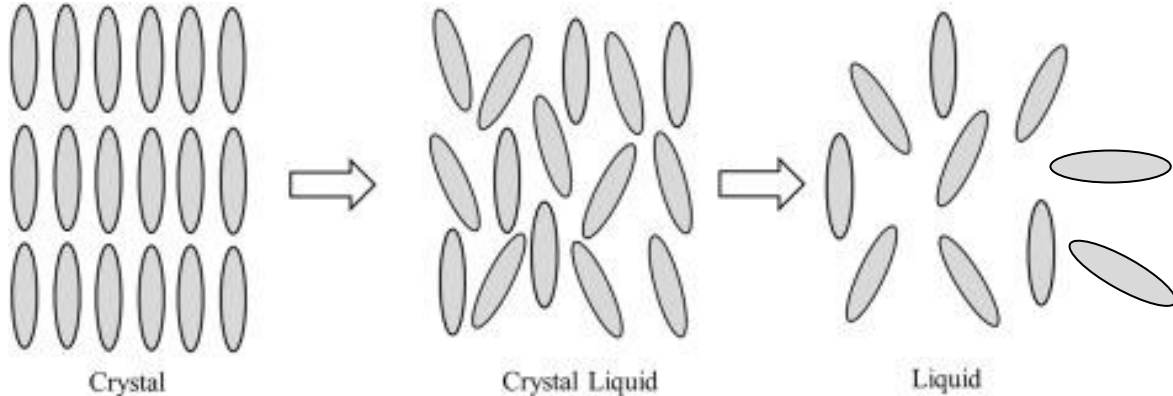
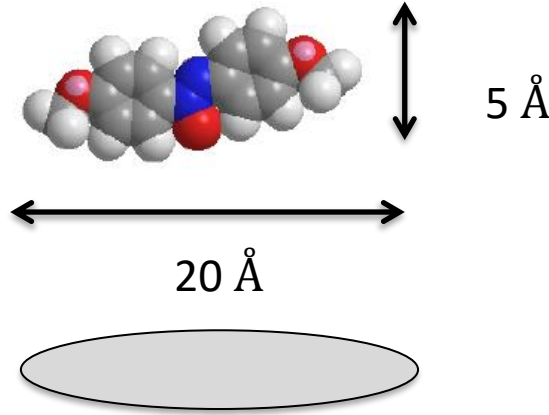
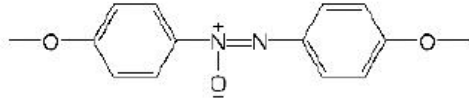


Liquid

- Crystalline properties: anisotropy of optical, electrical, and magnetic properties, as well as a periodic arrangement of molecules.
- Liquid properties: fluidity and the inability to support shear, formation and coalescence of droplets.

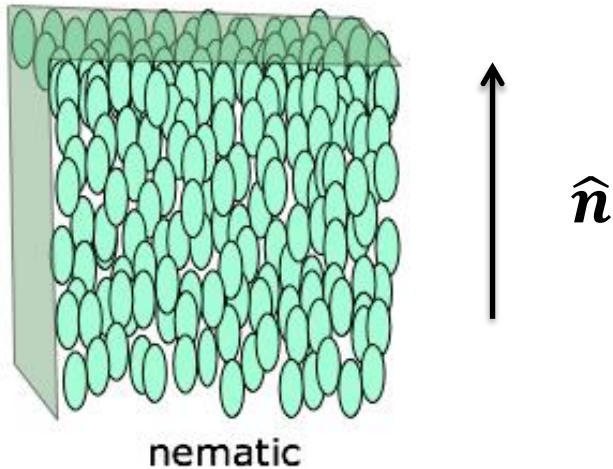
Cristais líquidos

PAA

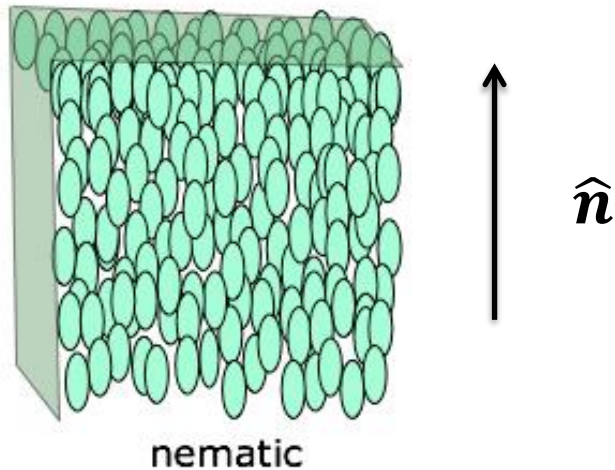


- Crystalline properties: anisotropy of optical, electrical, and magnetic properties, as well as a periodic arrangement of molecules.
- Liquid properties: fluidity and the inability to support shear, formation and coalescence of droplets.

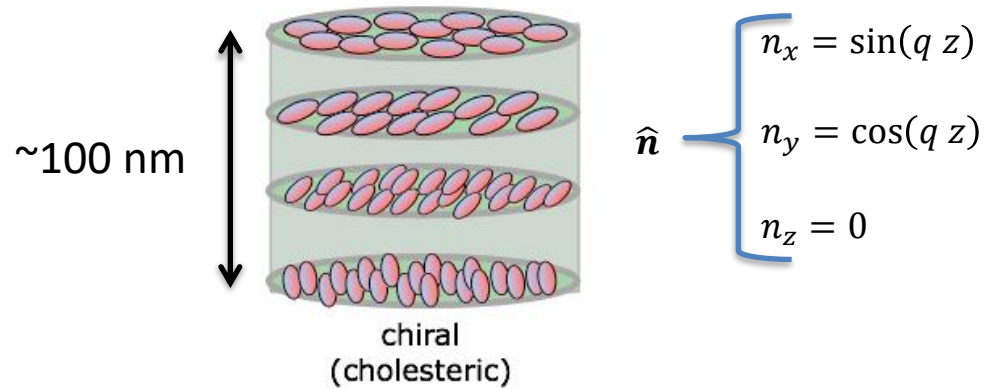
- **Nemáticos:** consiste de um conjunto de bastões cujo centros estão arranjados aleatoriamente, como partículas em um líquido. Isto é, não há ordem de longo alcance posicional. Porém, há ordem de longo alcance orientacional.



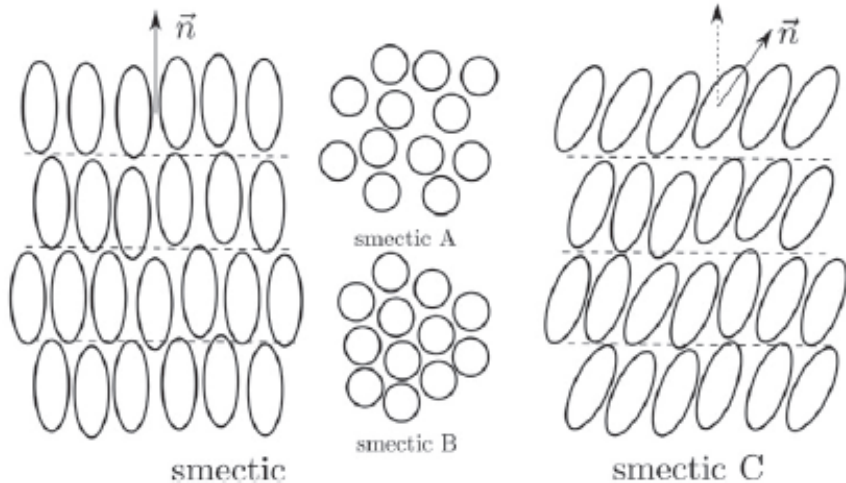
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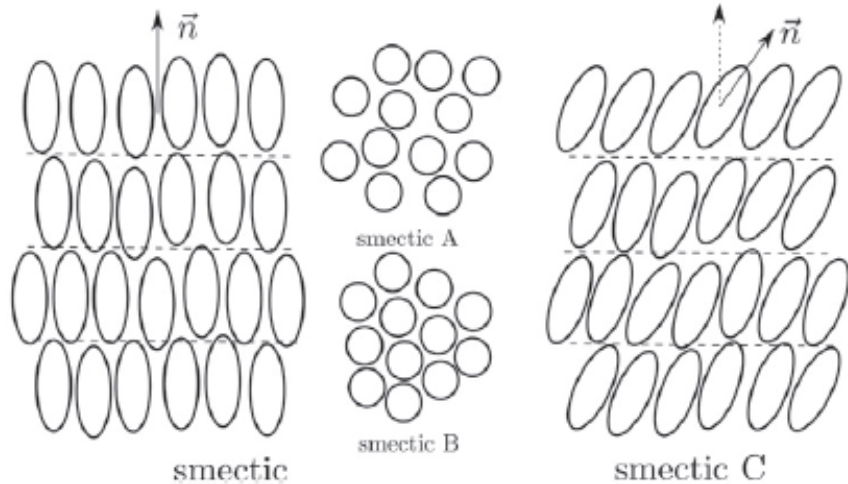
- **Colestéricos:** Fase similar à nemática, i.e. com centros arranjados aleatoriamente, mas com ordem de longo alcance orientacional. Porém, o vetor diretor muda de orientação ao longo de um dado eixo.



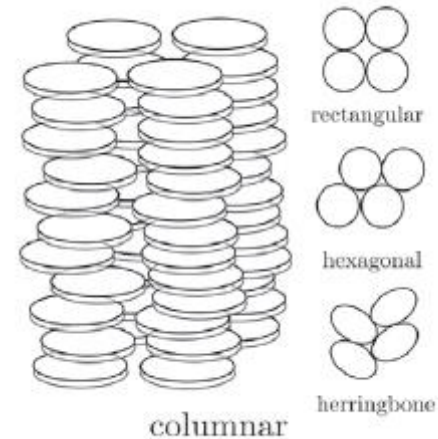
- **Esméticos:** nesta fase, os centros dos bastões possuem ordem posicional em uma dada direção (estratificação), além de ordem orientacional. Mas, as posições dos centros dos bastões nas camadas estratificadas é aleatória.



- **Esméticos:** nesta fase, os centros dos bastões possuem ordem posicional em uma dada direção (estratificação), além de ordem orientacional. Mas, as posições dos centros dos bastões nas camadas estratificadas é aleatória.



- **Colunares:** nesta fase, as moléculas são como discos e se organizam em colunas.



Displays de Cristais líquidos



LCD Display



OLED Display



AMOLED Display

Displays de Cristais Líquidos



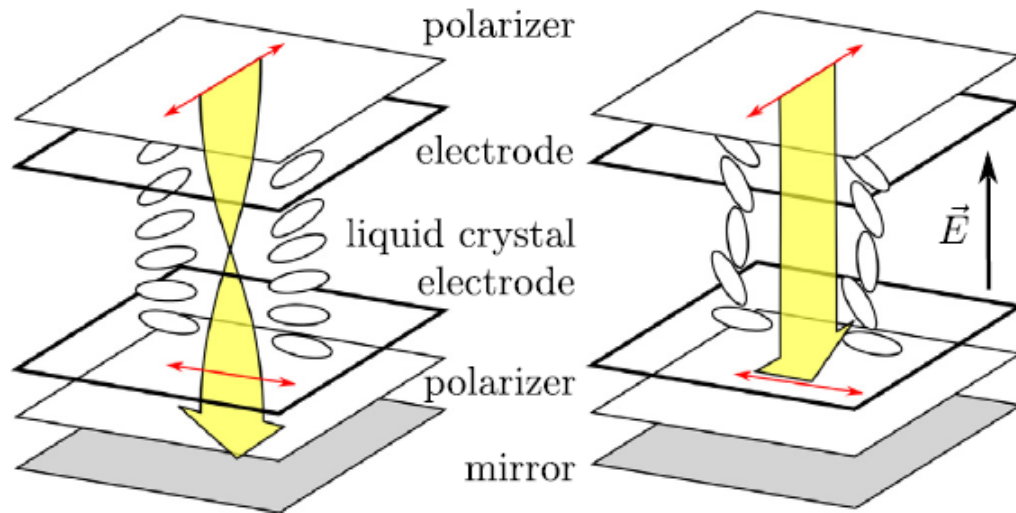
LCD Display



OLED Display



AMOLED Display



Displays de Cristais Líquidos



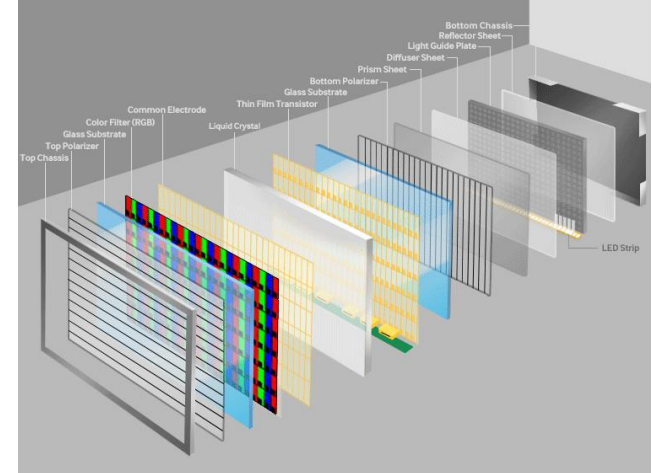
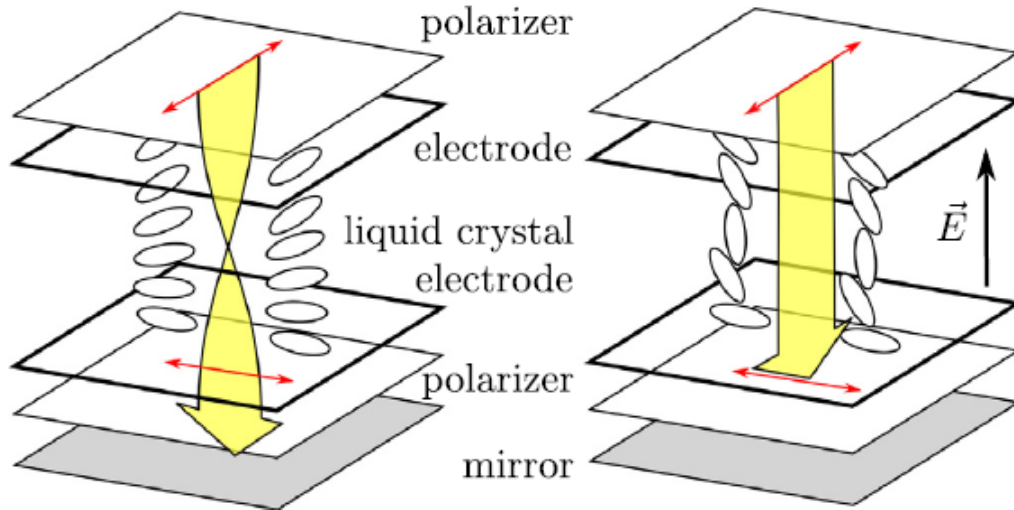
LCD Display



OLED Display



AMOLED Display



The Nobel Prize in Physics 1991

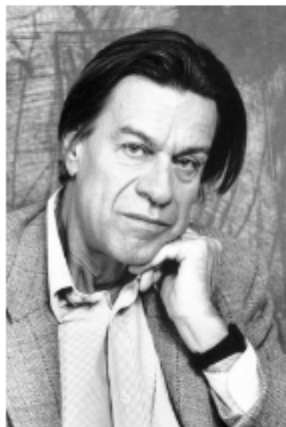


Photo from the Nobel
Foundation archive.

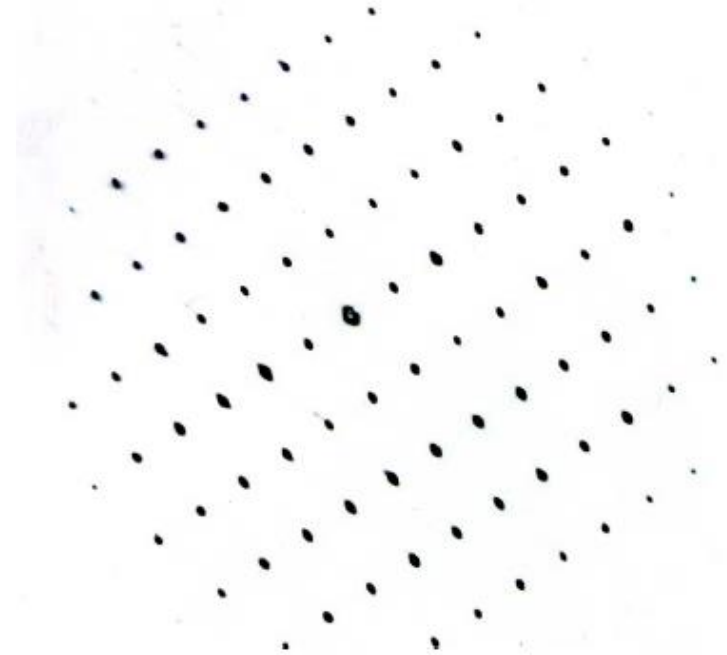
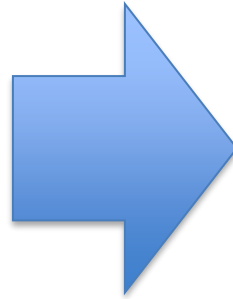
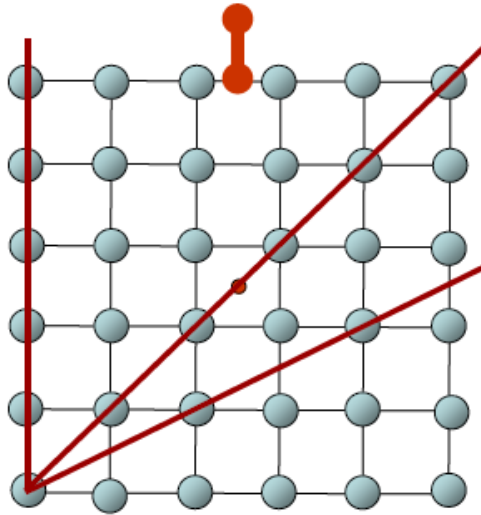
**Pierre-Gilles de
Gennes**

Prize share: 1/1

The Nobel Prize in Physics 1991 was awarded to Pierre-Gilles de Gennes "for discovering that methods developed for studying order phenomena in simple systems can be generalized to more complex forms of matter, in particular to liquid crystals and polymers"

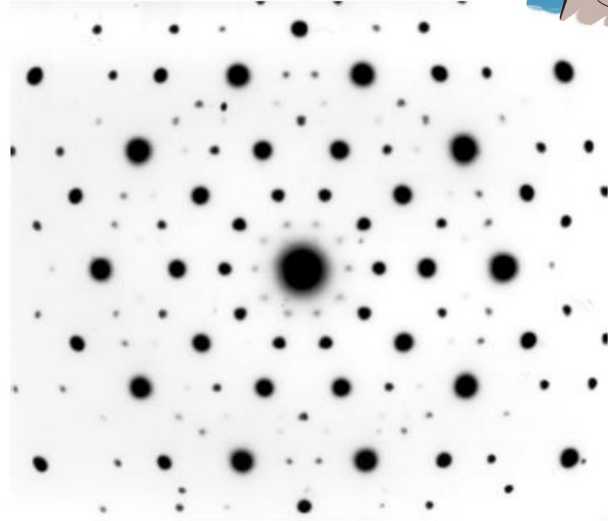
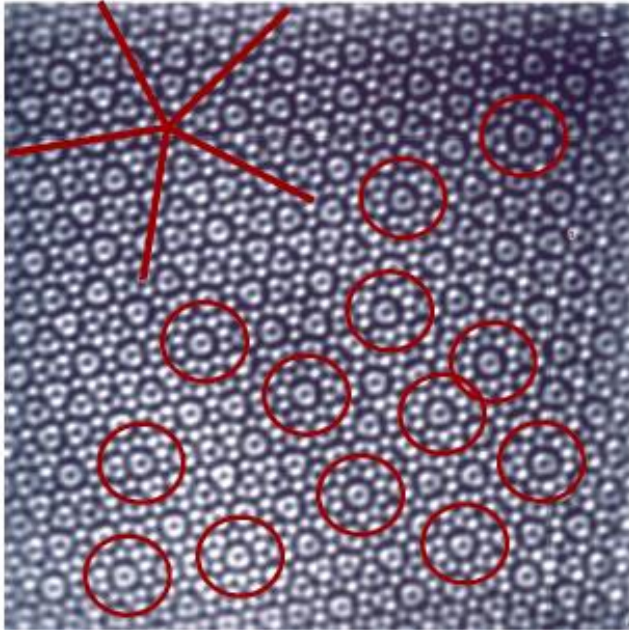
Quase-cristais

- Quase-cristais



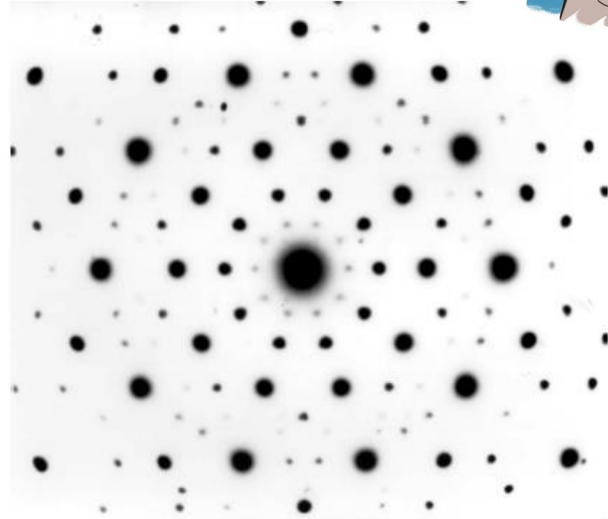
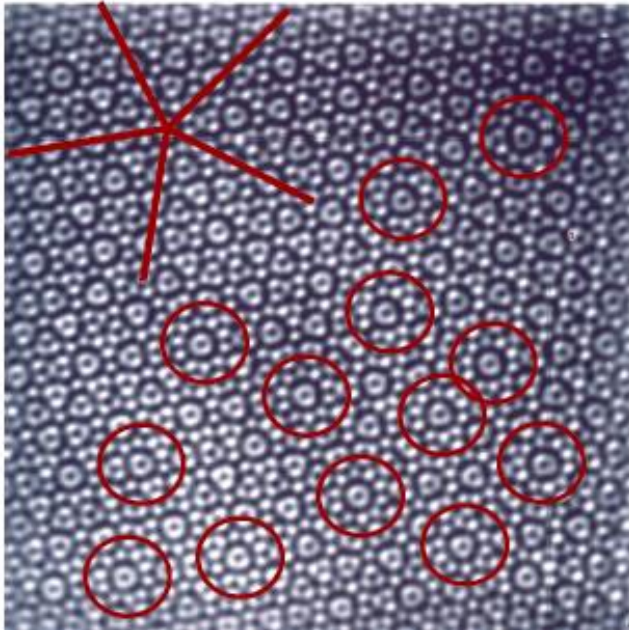
Quase-cristais

- Quase-cristais (*Dan Shechtman*)



Quase-cristais

- Quase-cristais (*Dan Shechtman*)

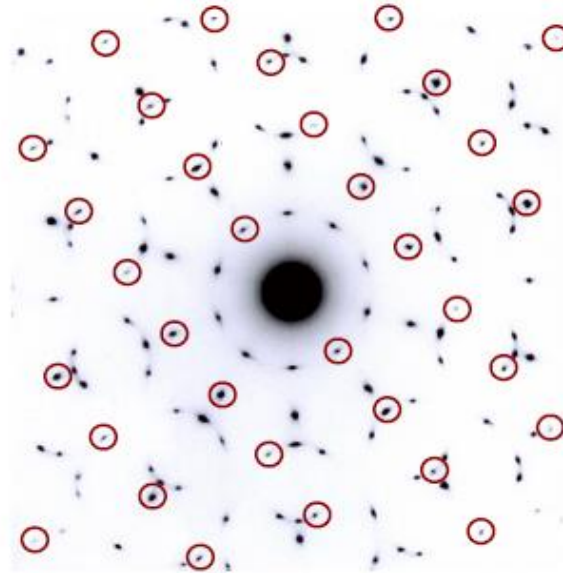
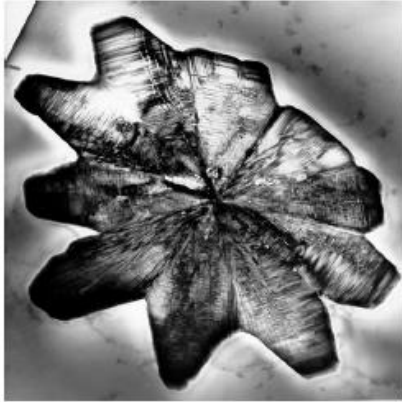


Razão da distância do ponto central aos demais

$$\tau = \frac{1 + \sqrt{5}}{2} \approx 1.618...$$

Quase-cristais

- Quase-cristais



Possibilidade descartada!!!

Quase-cristais

- Sequência de Fibonacci

$$a_1 = 1$$

$$a_2 = 1$$

$$a_3 = 2$$

$$a_4 = 3$$

$$a_5 = 5$$

$$a_6 = 8$$

$$a_7 = 13$$

$$a_8 = 21$$

$$a_{i+1} = a_i + a_{i-1}$$

$$\tau = \lim_{i \rightarrow \infty} \frac{a_{i+1}}{a_i}$$

Quase-cristais

- Sequência de Fibonacci

$$\begin{aligned}a_1 &= 1 \\a_2 &= 1 \\a_3 &= 2 \\a_4 &= 3 \\a_5 &= 5 \\a_6 &= 8 \\a_7 &= 13 \\a_8 &= 21\end{aligned}$$

$$a_{i+1} = a_i + a_{i-1}$$

$$\tau = \lim_{i \rightarrow \infty} \frac{a_{i+1}}{a_i}$$

- Sólido de Fibonacci

Regra:
A \rightarrow AB
B \rightarrow A

Quase-cristais

- Sequência de Fibonacci

$$\begin{array}{rcl} a_1 & = & 1 \\ a_2 & = & 1 \\ a_3 & = & 2 \\ a_4 & = & 3 \\ a_5 & = & 5 \\ a_6 & = & 8 \\ a_7 & = & 13 \\ a_8 & = & 21 \end{array} \quad \longrightarrow \quad B$$

- Sólido de Fibonacci

Regra:
A \rightarrow AB
B \rightarrow A

$$a_{i+1} = a_i + a_{i-1}$$

$$\tau = \lim_{i \rightarrow \infty} \frac{a_{i+1}}{a_i}$$

Quase-cristais

- Sequência de Fibonacci

$$\begin{array}{lcl} a_1 & = & 1 \\ a_2 & = & 1 \\ a_3 & = & 2 \\ a_4 & = & 3 \\ a_5 & = & 5 \\ a_6 & = & 8 \\ a_7 & = & 13 \\ a_8 & = & 21 \end{array} \quad \begin{array}{l} \longrightarrow \text{B} \\ \longrightarrow \text{A} \end{array}$$

- Sólido de Fibonacci


Regra:
A \rightarrow AB
B \rightarrow A

$$a_{i+1} = a_i + a_{i-1}$$

$$\tau = \lim_{i \rightarrow \infty} \frac{a_{i+1}}{a_i}$$

Quase-cristais

- Sequência de Fibonacci

$$\begin{array}{rcl} a_1 & = & 1 \\ a_2 & = & 1 \\ a_3 & = & 2 \\ a_4 & = & 3 \\ a_5 & = & 5 \\ a_6 & = & 8 \\ a_7 & = & 13 \\ a_8 & = & 21 \end{array}$$


- Sólido de Fibonacci

Regra:
A -> AB
B -> A

$$a_{i+1} = a_i + a_{i-1}$$

$$\tau = \lim_{i \rightarrow \infty} \frac{a_{i+1}}{a_i}$$

Quase-cristais

- Sequência de Fibonacci

a_1	= 1	→	B
a_2	= 1	→	A
a_3	= 2	→	AB
a_4	= 3	→	ABA
a_5	= 5		
a_6	= 8		
a_7	= 13		
a_8	= 21		

- Sólido de Fibonacci

Regra:
A -> AB
B -> A

$$a_{i+1} = a_i + a_{i-1}$$

$$\tau = \lim_{i \rightarrow \infty} \frac{a_{i+1}}{a_i}$$

Quase-cristais

- Sequência de Fibonacci

a_1	= 1	→	B
a_2	= 1	→	A
a_3	= 2	→	AB
a_4	= 3	→	ABA
a_5	= 5	→	ABAAB
a_6	= 8		
a_7	= 13		
a_8	= 21		

- Sólido de Fibonacci

Regra:
A -> AB
B -> A

$$a_{i+1} = a_i + a_{i-1}$$

$$\tau = \lim_{i \rightarrow \infty} \frac{a_{i+1}}{a_i}$$

Quase-cristais

Regra:
A -> AB
B -> A

- Sequência de Fibonacci

a_1	= 1	→
a_2	= 1	→
a_3	= 2	→
a_4	= 3	→
a_5	= 5	→
a_6	= 8	→
a_7	= 13	→
a_8	= 21	→

- Sólido de Fibonacci

B
A
AB
ABA
ABAAB
ABAABABA
ABAABABAABAAB
ABAABABAABAABAABAABA

$$a_{i+1} = a_i + a_{i-1}$$

$$\tau = \lim_{i \rightarrow \infty} \frac{a_{i+1}}{a_i}$$

Quase-cristais

- Sequência de Fibonacci

a_1	= 1	→
a_2	= 1	→
a_3	= 2	→
a_4	= 3	→
a_5	= 5	→
a_6	= 8	→
a_7	= 13	→
a_8	= 21	→

- Sólido de Fibonacci

B
A
AB
ABA
ABAAB
ABAABABA
ABAABABAABAAB
ABAABABAABAABAABAABA

Regra:
A -> AB
B -> A

(1 A e 1 B)
(2 A's e 1 B)
(2 A's e 1 B)
(5 A's e 3 B's)
(8 A's e 5 B's)
(13 A's e 8 B's)

$$a_{i+1} = a_i + a_{i-1}$$

$$\tau = \lim_{i \rightarrow \infty} \frac{a_{i+1}}{a_i}$$

A razão entre o #A e o #B converge para τ

Quase-cristais como um cristal em dimensões maiores: “cut-and-project construction”

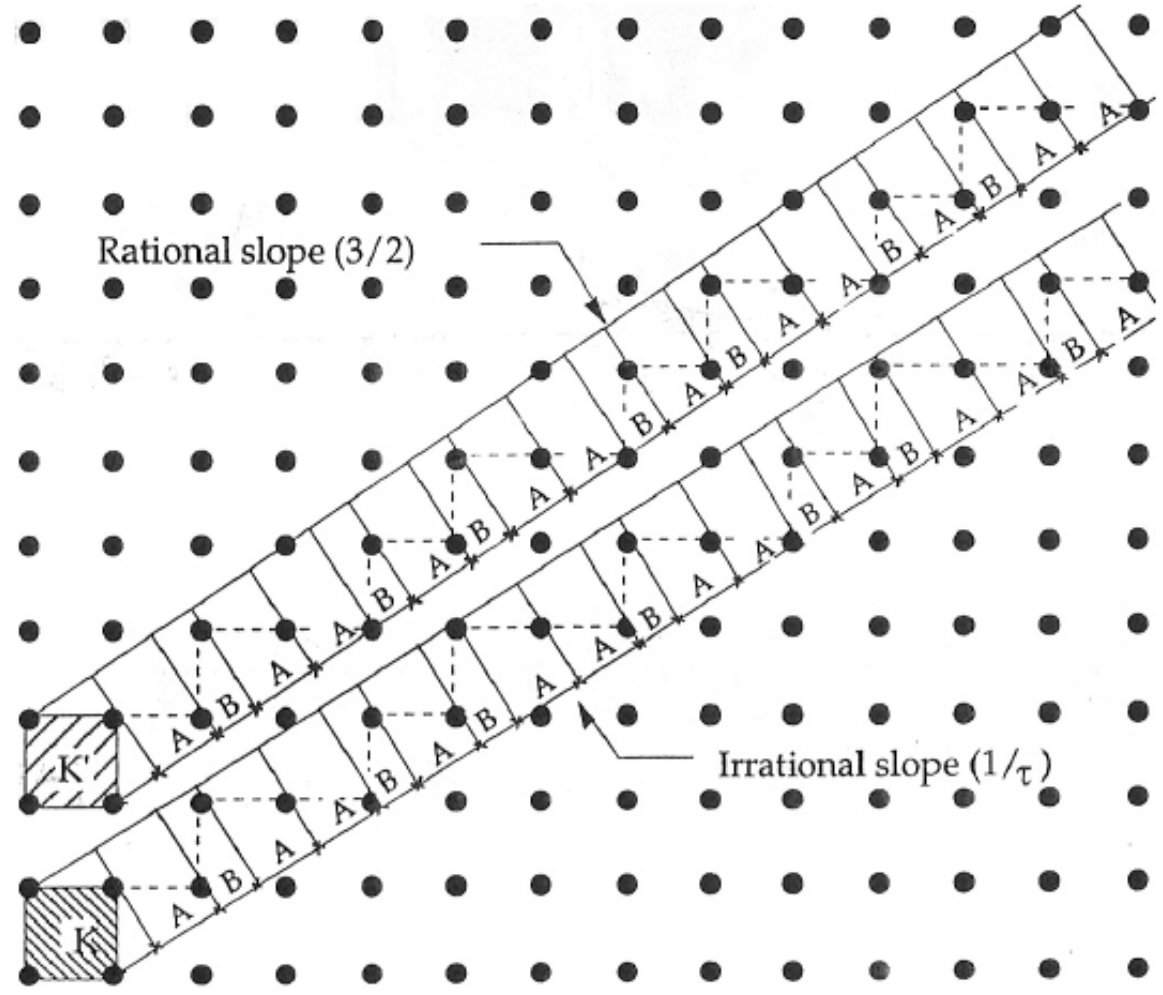
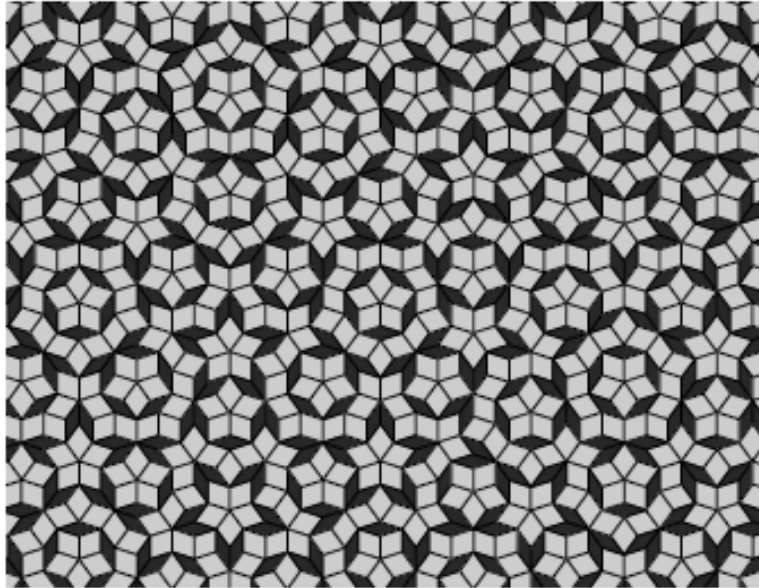


Table 1. List of the substitution rules determining the sequences usually considered in the study of self-similar aperiodic lattices.

Sequence	Set L	Substitution rule
Fibonacci	$\{A, B\}$	$g(A) = AB \quad g(B) = A$
Thue–Morse	$\{A, B\}$	$g(A) = AB \quad g(B) = BA$
Period-doubling	$\{A, B\}$	$g(A) = AB \quad g(B) = AA$
Silver mean	$\{A, B\}$	$g(A) = AAB \quad g(B) = A$
Bronze mean	$\{A, B\}$	$g(A) = AAAB \quad g(B) = A$
Copper mean	$\{A, B\}$	$g(A) = ABB \quad g(B) = A$
Nickel mean	$\{A, B\}$	$g(A) = AB BB \quad g(B) = A$
Triadic Cantor	$\{A, B\}$	$g(A) = ABA \quad g(B) = BBB$
Rudin–Shapiro	$\{A, B, C, D\}$	$g(A) = AC \quad g(B) = DC \quad g(C) = AB \quad g(D) = DB$
Paper folding	$\{A, B, C, D\}$	$g(A) = AB \quad g(B) = CB \quad g(C) = AD \quad g(D) = CD$

Sólidos desordenados

- Quase-cristais



Mosaico de Penrose

The Nobel Prize in Chemistry 2011



© The Nobel Foundation. Photo:
U. Montan

Dan Shechtman

Prize share: 1/1

The Nobel Prize in Chemistry 2011 was awarded to
Dan Shechtman "for the discovery of
quasicrystals"

AGL: So basically there was the moment in 1982 and the moment when the paper came out. What happened in between?

DS: In the end of 1983, I left NBS to go back to Israel, and I started to talk about my 10-fold diffraction patterns and something new in crystallography. And people said this is nonsense. So the reaction varied, between the reaction of my host, John Cahn, who was positive, and said "Danny this material is telling us something and I challenge you to find out what it is". He encouraged me to continue studying it. The other side, the worst reaction, was from my group leader who came to my office one day, smiling sheepishly and put a book on my desk on x-ray crystallography. He told me to "read this book and you'll understand what you're talking about". I said "I don't need to read the book. I'm a professor at the Technion. My material is not in the book". A few days later, the group leader said to me "you are a disgrace. I want you to leave. I cannot have my name associated with you." So I had to leave my group but I found another researcher there who was willing to adopt a scientific orphan- me.

AGL: What advice do you offer young scientists about dealing with rejection?

DS: I have several [pieces of advice]. Number one, become an expert in something you like. Try to be the best in something you like. Once you are an expert and someone criticizes you, then you can say "you may be the greatest scientist in the world but I am an expert in this". Number two, pay attention to details, especially surprising details that you don't expect. And if you find something strange, don't let it disappear. Study and find out what it is. Sometimes it will be an artifact, but in some cases, you've made a great discovery that will determine your success in science and your career. So be like a Rottweiler: bite and don't let go! And if somebody says this is rubbish, say "don't tell me it's not in the book. Show me what's wrong". Check yourself ten times before you start talking. Make sure you don't make a mistake. But as an expert, trust yourself.

<https://www.aps.org/publications/apsnews/201703/shechtman.cfm>

